1 High-performance polylactic acid compressed strawboard using pre-treated and functionalised

2 wheat straw

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13 Abstract

- An eco-friendly pre-treatment coupled with surface functionalisation were developed to enhance the 14 quality of wheat straw particles to be used for development of high-performance polylactic acid (PLA) 15 compressed strawboard. Eco-friendly hybrid pre-treatment (i.e., hot water followed by steam, H+S) and 16 17 surface functionalisation processes employing attapulgite nanoclay (AT) and graphene nanoplatelets (G) were used to obtain an appropriate wheat straw surface quality while increasing its compatibility with 18 the PLA matrix. The successful pre-treatment and surface functionalisation of wheat straw particles was 19 20 verified through characterisation techniques, including SEM, FTIR, XRD, Raman spectroscopy, and TGA. Tensile strength and water absorption properties of compressed strawboards were examined to 21 22 investigate the influence of pre-treatment and surface functionalisation of wheat straws. The maximum 23 tensile strengths of 28 MPa and 27 MPa were recorded for 10H+S-AT and 10H+S-G samples, respectively, 24 which are considerably higher than the value (i.e., 9.7 MPa) registered for the sample without pretreatment and surface functionalisation (i.e., 10UN). The lowest water absorption after 24 h of 25 immersion was registered for 10UN-G (i.e., 1.6%), which is 11% and 31% lower than the 10UN and 10H+S 26 samples, respectively. The effect is attributed to an improved interfacial bond between wheat straw and 27 PLA matrix due to the graphene surface functionalisation, as evidenced by the SEM. 28
- Keywords: Polylactic acid compressed strawboards; wheat straw; attapulgite nanoclay; graphene
 nanoplatelets; pre-treatment; surface functionalization.
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32 **1. Introduction**

Polylactic acid (PLA) is among the most prominent biodegradable polymers and has considerable ability 33 of replacing petroleum-based plastics. PLA is made from fermented lactic acid derived from renewable 34 35 agricultural sources such as corn and wheat. PLA has a broader range of applications than other biodegradable polymers due to its superior mechanical characteristics, biocompatibility and 36 biodegradability, ease of processing, and thermal stability (Gupta et al., 2007; Tawakkal et al., 2014). 37 38 However, PLA is far more expensive than its petroleum-based competitors, severely limiting its 39 commercial application in the industrial sector (Qin et al., 2011). Recent research suggests replacing the PLA matrix with low-cost agricultural waste components to alleviate this problem. Various types of 40 41 agricultural waste materials, including hemp and jute strands, corn cobs, and rice husks have received a lot of research attention as a possible eco-friendly replacements in bio-based composites (Liu et al., 42 2017; Melo et al., 2014; Tribot et al., 2018). Wheat straw, a renewable agricultural biomass with chemical 43 constituents of cellulose, hemicellulose, lignin, and extractives, has the potential to successfully 44 substitute polymer matrix in a variety of applications. Nevertheless, its high concentration of 45 hydrophobic elements, i.e., waxy cuticle layers, inorganic silica, and extractives, results in inadequate 46 surface characteristics, impairing the interfacial bond quality between the wheat straw and polymer 47 48 binder (Ghaffar et al., 2017a; Ghaffar and Fan, 2017, 2015).

49 Pre-treatment and coating processes have been introduced as a feasible approach to overcome these shortcomings in numerous efforts (Ghaffar et al., 2017b; Ghaffar and Fan, 2015; Hýsková et al., 2020). 50 To improve adhesion between biomass and polymer matrix and, as a result, the ultimate performance 51 of bio-based composites, a variety of pre-treatment solutions have been developed, which can be 52 classified into physical and chemical methods. When evaluating sustainability, it should be highlighted 53 54 that although chemical pre-treatments substantially modify biomass structure, they may not be 55 economically feasible, and the required chemicals could be hazardous. Thus their environmental 56 implications would be unfavourable compared to that of physical pre-treatments (Chougan et al., 2020; 57 Fan et al., 2018). For instance, pre-treatment techniques such as alkaline chemical, microwave, boiling and steaming, and enzymatic pre-treatments were used to enhance the performance of bio-based PLA 58 59 composites (Laadila et al., 2017). In addition, coupling agents such as silane coupling agent were used to

60 improve the compatibility interface and thermal stability performance of wheat straw/PLA composites

61 (Chen et al., 2021).

62 Depending on the size and origin of the filler biomass resources, bio-based PLA composites have been 63 used in various applications. Composites using micron-sized biomass materials could be a valuable source for developing single-use plastics, cutlery, and food packaging. To eliminate the use of chemicals 64 in food-grade applications, Mousa et al. (2022) used biomass filler derived from date palm rachis 65 without any chemical treatments or surface modification (Mousa et al., 2022). In addition, reduced 66 67 density, fire resistance, acoustic emission resistance, low cost, availability, high energy efficiency, and an appropriate modulus-weight ratio makes it suitable for non-load bearing applications in the construction 68 69 and automotive industries (Das et al., 2022; Pawar, K.S., Bagha, A.K., Bahl, S., Nandan, 2022; Prakash et al., 2022) 70

71 This study utilised a selective separation of wheat straw nodes that serve as defects in bio-based 72 composites to optimize the bonding efficiency between wheat straw particles and polymer matrix. Thereupon, a light physical pre-treatment (H+S) was employed to improve the surface properties of 73 wheat straw particles. This pre-treatment method was chosen based on the author's previous study in 74 which H+S demonstrated a positive effect on the extraction of undesirable chemicals and improvements 75 76 in tensile strength (Chougan et al., 2020). Graphene nanoparticles and attapulgite nanoclay at 0.1 wt.-% 77 and 1 wt.-% weight percent, respectively, were used to functionalise the surfaces of both untreated and 78 pre-treated straws. The author's previous study investigated the same surface functionalisation system, which confirms the nanomaterial's surface modification role in improving the interfacial bonding 79 between cementitious composite matrix and straw particles (Chougan et al., 2022). To the best of the 80 authors' knowledge, no studies on wheat straw has implemented the pre-treatment coupled with 81 82 surface functionalisation with graphene and attapulgite nanoparticles to be employed in bio-based 83 polymer composites. Extensive material characterisations, including FTIR, SEM, Raman, XRD, and TGA, 84 were carried out to confirm the adequate surface functionalisation of wheat straw particles. Bulk 85 property evaluation of PLA compressed strawboard was also performed in order to assess the impacts of surface functionalisation and pre-treatment processes on potential enhancements of mechanical 86 87 performance and dimensional stability.

88 The testing procedure and investigation strategy for this research is depicted in **Figure 1**.

- 89
- 90 Here **Figure 1.** Schematic structure of experimental testing programme.

91 **2. Materials and method**

92 Thermoplastic polylactic acid (PLA) used as matrix in this work was purchased from Tecnaro GmbH's 93 Ilsfeld, Germany. Wheat straw biomass (Triticum aestivum L.) was supplied from a residential farm (Middlesex, UK) harvested in late summer 2019. Two different coupling agent nanomaterials, were 94 95 employed for surface functionalisation, namely, attapulgite nanoclay (AT) supplied by Lawrence 96 Industries Ltd., UK, and graphene nanoplatelets (G), provided by Nanesa S.r.l., Italy. Microstructure images of G, AT are presented in Figure 2 (a and b). A fringed shape with a wrinkling surface was 97 98 highlighted for isolated graphene nanoplatelets. These particles accumulate into large agglomerates, as 99 seen by the distinctive shape of specimens produced from expanded graphite flakes. AT particles, however, exhibit a rough surface and an angular morphology with sharp edges. The detailed 100 101 characterization of both G and AT particles was presented in the author's previous studies (Chougan et 102 al., 2021; Lamastra et al., 2021). Based on the supplier's information (i.e., Nanesa S.r.l. for graphene, and 103 Lawrence Industries Ltd. for attapulgite), the industrial-scale cost of functionalising agents is 104 approximately 70 €/kg and 6 €/kg for graphene and attapulgite, respectively. Consequently, the 105 estimated production cost of strawboard of size 1 m³ with 20 vol.% straw replacement will be increased around € 5.5 and €4.5 by the incorporation of 0.1% graphene and 1% attapulgite particles. It should be 106 107 noticed that due to scientific research in our study the attapulgite clay was obtained from chemical 108 company therefore its price was high. However, there are many nanoclay deposits available globally as 109 this is naturally occurring material, thus it's price could decreased when the material is commercialised.

110 **2.1.** Wheat straw preparation, pre-treatment, and functionalisation

Prior to the pre-treatment and surface functionalization processes, as-received wheat straw was cleaned and oven-dried at 100 \pm 5 °C for 24 hours (**Figure 3-i**). Based on the author's previous work (Ghaffar and Fan, 2015), due to inconsistent shape and chemical functional groups distribution throughout the straw stem, only the internode was used for bio-based composites. Straw internode sections were shredded to obtain straw particles using a Retsch SM 100 cutting mill. More than 99% of the straw particles were found to be in the range of 65 – 2000 µm (see **Figure 2 c**). 117 An eco-friendly and hazardous-substance-free pre-treatment combining hot water (H) and steaming (S) 118 was carried out on wheat straw particles (Figure 3-ii). Throughout the (H) stage, straw particles were 119 introduced into a pressure cooker for 60 minutes at a constant pressure of approximately 0.1 MPa. Then, 120 boiled-straw particles were steamed immediately for another 30 minutes using a mesh basket positioned 121 directly above boiling water. Both pre-treated (H+S) and untreated (UN) straw particles were used in order to investigate the effect of surface functionalisation. Before surface functionalisation stage, to 122 123 achieve an effective dispersion of attapulgite nanoclay (AT) and graphene nanoplatelets (G), an ultra-124 sonication technique in an aqueous solution was performed. The ultrasonic time and power were set to 90 minutes and 200 W/cm², respectively. In this study, the optimum concentrations of 0.1 wt.% and 1 125 wt.% were utilised for AT and G, respectively, which have been reported in previous investigations 126 (Scaffaro et al., 2020; Zhu et al., 2019). Correspondingly, (UN) and (H+S) straw particles were introduced 127 128 in pre-dispersed G and AT aqueous solutions. The straw particles and pre-dispersed solution were stirred 129 for 12 h at 80 °C using a hot plate and magnetic stirrer until 90% water was evaporated from the solution (Figure 3-iii). Each sample was then oven-dried at 100 °C for 24 h, where the AT and G particles stick to 130 131 the outer surface of UN and H+S straw particles. More details on surface functionalisation and pretreatment procedures can be found in the author's previous research papers (Chougan et al., 2022, 132 133 2020). The authors believe that the proposed pre-treatment (i.e., H+S) could limit energy use as well as involvement of chemical agents in comparison to bleaching, alkaline oxidation, and plasma. It is 134 acknowledged that the wastewater generated by the H+S pre-treatment could contain hazardous 135 substances, however compared to the chemical pre-treatments and disposal of agricultural waste 136 materials in the environment (e.g., burning), the H+S is relatively considered as eco-friendly. Moreover, 137 as per author's previous study, it is evident that the extractives in nodes are 10–15% higher than in 138 139 internodes. Therefore, removing node section before pre-treatment guarantees the minimum release 140 of these substances in the environment (Ghaffar and Fan, 2017). It is worth noting that based on the 141 author's experience, all the wastewater in the boiling stage of the pre-treatment will evaporate during 142 the steaming stage.

Here→ Figure 2. Microstructure profile of (a) graphene nanoplatelets, (b) attapulgite nanoclay and (c)
 particle size distribution analysis of wheat straw particles

145 Here **Figure 3**. Schematic framework of different stages of wheat straw board preparation.

146 **2.2.** Compressed strawboard sample preparation

147 A total of 30 compressed strawboards were fabricated using H+S and UN straw particles and their surface 148 functionalised derivatives, i.e., UN-G, UN-AT, H+S-G, and H+S-AT. PLA polymer pellets were replaced by wheat straw particles with volume ratios of 10, 20, 30, 40, and 50 (see Table 1). For each composition, 149 150 straw particles were dry-mixed with polymer pellets to ensure the uniform distribution of materials. The mixture was placed into the steel mould of 100 mm x 100 mm x 20 mm and heated in the hot-air oven 151 for 15 minutes at 180 ± 10 °C to facilitate the PLA melting process. After the pre-heating step, straw 152 particles were mixed with polymer pellets to ensure uniform distribution, the mould with softened PLA 153 and straws was then hot-pressed (see Figure 3-iiii) for 20 min at 180 °C under the pressure of 10 MPa. 154 155 Samples were subjected to a 15 kg weight immediately after removing the moulds from the hot-press to prevent the samples from expanding. Subsequently, the resulting bio-composite was allowed to cool 156 157 down for 20 minutes to reach room temperature. Finally, specimens for the tensile test were taken by 158 cutting the bio-composites into 20 mm wide strips.

159 Here **Table 1.** Mix formulations for PLA compressed strawboard manufacturing

160 **2.3.** Characterisations and testing of functionalised wheat straw

161 **2.3.1. Microstructure**

Scanning electron microscopy (SEM, VEGA3 TESCAN) was used to examine the surface morphology and microstructure of functionalised wheat straw samples and compare them to control specimens. Ten straw particles with a size of 5mm³ were examined for each composition to observe the surface alterations induced as a result of surface functionalisation. A total of ten specimens with an approximate size of 10 mm³ were taken from fractured sections of composite samples in tensile testing to analyse the interfacial bonding between the functionalised straw particles and polymer matrix. Samples were chromium-coated to ensure appropriate electrical conductivity.

169 2.3.2. Raman spectroscopy

- In order to characterise the presence of functionalising agents on the surface of straw particles, Raman
 spectroscopy assessments were carried out at an ambient temperature in the Raman shift range of 800
 1800 cm⁻¹ using inVia Raman Microscope (Renishaw), equipped with a 785 nm wavelength laser beam.
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175 2.3.3. X-Ray diffraction

The X-ray diffraction (XRD) analysis was performed to examine the mineralogical composition of the wheat straw particles before and after pre-treatment and surface functionalisation processes by means of an Aeris Diffractometer (Bruker) with Cu-Kα radiation, 20 of 5–90°, at 40 kV and 40 mA, and a wavelength of 1.542 Å. The crystallinity index (CI) was determined using an empirical technique provided by Segal et al. (1959) (**Eq.1**) based on the XRD data (Segal et al., 1959). The CI term is only valid for comparison purposes since it describes the order of crystallinity instead of the crystalline areas' crystallinity.

$$CI(\%) = \frac{I_{002} - I_{amorph}}{I_{002}} \times 100$$
 Eq.1

183 Where: I_{002} is the maximum intensity of the (002) lattice diffraction and reflects both the crystalline and 184 amorphous regions of the material (i.e., 20 between 20° and 24°), and I_{amorph} implies the amorphous area 185 (i.e., 20 between 17° and 20°).

186 **2.3.4.** Thermogravimetric analysis

The effect of surface functionalisation on the thermal degradation of straw particles was investigated on
 3–4 mg of straw samples using thermogravimetric analysis (TGA). TGA analysis was conducted via TA
 Instrument SDT Q600 under airflow with a heating range of 20 to 650 °C at a rate of 10 °C/min.

190 **2.4.** Mechanical properties

A total of 5 strawboard strips of size 100 mm x 20 mm x 20 mm were utilised to evaluate the tensile strength performance of each composition and mean values were taken as a representative. Tensile strength tests were performed employing an Instron 5969 universal testing machine equipped with wedge action tensile grips as per ASTM D3039/D3039 M.

195 **2.5.** Water absorption of bio-based composites strawboards

As per BS 5669–1:1989, the dimensional stability of bio-based composites, i.e., water absorption (WA), was measured on a batch of six samples of size $100 \times 20 \times 20 \text{ mm}^3$. The test was performed by submerging the samples in water at a constant temperature of 20 ± 2 °C. After 2 h and 24 h of exposure to the water, the weight of each sample was precisely measured using a digital scale with an accuracy of 0.01 (g). The water absorption percentages were determined as follows (**Eq.2**). A batch of three samples have been evaluated for each composition and the average results are reported.

$WA(\%) = \left[\frac{W_2 - W_1}{W_1}\right] \times 100$ Eq.2	
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202 Where: WA is water absorption (%), W_1 (g) and W_2 (g) are the weights of samples before and after water 203 immersion, respectively.

204 3. Result and discussion

3.1. Surface and microstructural changes due to pre-treatment and fuctionalisation

The microstructure of the wheat straw's cross-section before and after (H+S) pre-treatment were 206 207 examined and using SEM. As shown in Figure 4, H+S pre-treatment alternates the microstructure of 208 wheat straw particles in terms of (i) parenchyma expansion and (ii) epidermis thickness reduction. 209 Aforementioned desirable alternations lead to the deeper penetration of liquid PLA matrix inside the 210 straw's parenchyma which in turn leads to an intimate bonding and subsequently, improving the 211 strength of PLA compressed strawboard (Chougan et al., 2020; Ghaffar et al., 2017a). As presented in the author's previous work, an epidermis thickness reduction of about 47% occurred after (H+S) pre-212 treatment (Chougan et al., 2020). Reducing the size of the epidermal layer, which includes hydrophobic 213 214 silicone and waxes, could be advantageous for improving interfacial interaction between polymer matrix 215 and wheat straw owing to the lower hydrophobic contact area. Moreover, the results also indicated a 216 12% parenchyma expansion as a result of (H+S) pre-treatment, which guarantees a more in-depth penetration of the PLA matrix (see Figure 4). 217

Here→ Figure 4. SEM images of internode cross-section profile of (a) untreated (UN) and (b) pretreated (H+S) straw particles

220 As seen in Figure 5 and 6, SEM analysis was also performed on the surface of wheat straw before and after pre-treatment and surface functionalisation processes. The results indicated a continuous and 221 222 smooth surface morphology for H+S straw particles (see Figure 6 a and d), whereas UN samples showed a rough surface (see Figure 5 a and d). The microstructure comparison of surface functionalised and non-223 224 functionalised straw particles confirm that AT and G particles have been effectively attached to the 225 straw's surface. However, due to the lower hydrophobic contact area of the H+S samples, a higher 226 quantity of AT and G particles were presented on the pre-treated samples. In order to prove the 227 aforementioned statements, FTIR-ATR spectroscopy (Figure 7) was used to characterise the surface chemical distribution of pre-treated and untreated straw. The bands at 1595 cm⁻¹ and 1510 cm⁻¹ 228

229 represent the aromatic ring stretch of lignin (Yang et al., 2020). As shown in (Figure 7-i), the lignin peaks 230 for pre-treated straw samples was slightly attenuated than that of untreated straw samples, which 231 implies a partial lignin extraction as a result of H+S pre-treatment. The major components of extractives and hemicellulose are represented by mode at 1735 cm⁻¹, which encompasses carboxyl groups in the 232 acids and esters of acetic, p-coumeric, ferulic, and uronic acids (Alemdar and Sain, 2008a). As seen in 233 Figure 7-ii, comparing to UN straw, the intensity of this band was mitigated in the H+S samples. Peaks at 234 bands 2920 cm⁻¹ and 2850 cm⁻¹ show asymmetric and symmetric CH₂ stretching bands that correspond 235 to the aliphatic fractions of waxes (Saari et al., 2014). The results confirmed that H+S straw samples 236 contain lower frictions of wax and inorganic compounds than that of UN samples (see Figure 7-iii). This 237 238 suggests that pre-treatment reduces the waxes and inorganic chemicals on the surface of straw particles. As evident in Figure 5 (b and e) and Figure 6 (b and e), for both pre-treated and untreated samples, the 239 distinctive attapulgite nanoclay particles are effectively covering the straw surface. Figure 5 (c and f) and 240 Figure 6 (c and f), on the other hand, demonstrate isolated graphene nanoplatelets sticking to the straw 241 surface. In order to validate the surface functionalisation, Raman spectroscopy test was performed (see 242 Figure 8) as an intuitive method to investigate the presence of functionalising agents on the surface of 243 straw particles. Two typical characteristic peaks of graphene particles, i.e., D band at 1335 cm⁻¹, 244 245 attributable to the carbon lattice disorder specific of edges and defects in the aromatic structure, and G band at 1585 cm⁻¹, referring to the C sp2 in-plane vibration of the graphene lattice, were clearly observed 246 on the straw particles' surface (Owens, 2015). As evident in Figure 8, UN-G and H+S-G straw samples 247 both have typical characteristic peaks of graphene particles. Typical characteristic signals of attapulgite 248 nanoclay particles were also detected. For UN-AT and H+S-AT samples, the distinctive peaks of AT were 249 250 observed. None of the aforementioned characteristic peaks can be ascertained on neat, i.e., not 251 functionalised, UN and H+S straws (marked in blue and green, respectively), indicating that AT and G 252 particles homogeneously covered the straw particles.

- Here→ Figure 5. SEM images of (a and d) untreated (UN) straw particles and their composites with (b
 and e) AT nanoclays and (c and f) graphene nanoplatelets.
- Here→ Figure 6. SEM images of (a and d) pre-treated (H+S) straw particles and their composites with
 (b and e) AT nanoclays and (c and f) graphene nanoplatelets.

257 Here **Figure 7**. FTIR-ATR spectrum of untreated (UN) and pre-treated (H+S) straw particles.

Here→ Figure 8. Raman spectra of untreated, and pre-treated straw particles and their composites
 with G and AT nanoparticles.

260 **3.2.** Thermogravimetric analysis

261 Thermogravimetric analysis (TGA) was conducted on wheat straw particles before and after surface 262 functionalisation to validate their thermal stability, as shown in Figure 9. TGA analysis presented in the 263 thermograms, which revealed three distinct stages of thermal degradation. The initial stage of the straw sample's weight loss was observed at 100 - 150 °C owing to the moisture evaporation. The cellulose and 264 hemicellulose in the straw sample broke down at temperature range between 250 and 350 °C, resulting 265 in substantial weight loss. The decomposition of non-cellulosic components, particularly in the case of 266 267 graphene nanoplatelets, caused the ultimate weight loss in the temperature range of 500 - 600 ° C. 268 However, the aforementioned temperature range (i.e., 500-600 ° C) is insufficient to degrade AT 269 nanoclay particles (Ergudenler and Ghaly, 1992; Ghaffar and Fan, 2015). Several authors (Ghaffar and 270 Fan, 2015; Zandi et al., 2019) suggested a direct correlation between the onset of degradation 271 temperature and the thermal stability of straw, more specifically, higher onset of degradation indicates enhanced thermal stability of straw. As it can be seen in Figure 9-i, due to the extraction of waxy 272 273 compounds and hemicelluloses induced by the H+S pre-treatment, the onset of degradation 274 temperature for H+S pre-treated straw and its corresponded surface functionalised particles (i.e., 255 °C for H+S, 254 °C for H+S-G, and 276 °C for H+S-AT) are relatively higher than that of untreated straw 275 276 samples (i.e., 248 °C for UN, 253 °C for UN-G, 265 °C for UN-AT).

The high content of silica and ash silica on the surface of straw particles makes it challenging to incorporate these components into bio-based composites applications. According to Qin et al. (2011), the residual weight at 600 °C corresponds to the remaining ash content (Qin et al., 2011). Compared to the UN samples, the ash content of H+S samples is substantially lower. The reduction in the ash and silica content of the straw particles enhances the interfacial interaction between the straw and PLA matrix, therefore, improving the performance of PLA compressed strawboards.

Here→ Figure 9. TGA thermogram data of untreated and H+S pre-treated wheat straw particles with
 and without surface functionalisation.

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3.3. Pre-treatment and surface functionalisation effect on crystallinity index

288 Figure 10 exhibits XRD patterns of G, AT, and functionalised straw particles to analyse the crystalline 289 phases of specimens. All of them are typical for cellulosic materials. The maximum intensity was recorded at $2\theta = 21.55^{\circ}-22^{\circ}$, corresponding to the 002 lattices (Ghaffar and Fan, 2015; Kaushik et al., 290 2010). Furthermore, the existence of cellulose in the form of cellulose I crystal was confirmed by the 291 secondary peak around $2\theta = 18^{\circ}$. The crystallinity index of all straw samples is shown in **Table 2**. The 292 crystallinity index of UN straw particles was calculated to be 55.6 %. The crystallinity index increased by 293 294 approximately 3 % following H+S pre-treatment and reached 57.4 %. The extraction of fundamentally 295 amorphous polymers within the constituents of wheat straw, particularly lignin and hemicellulose, is 296 directly proportional to the mere increase in crystallinity index induced by the H+S pre-treatment (Alemdar and Sain, 2008b). According to Zhu et al. (2006), the rise in cellulose content is related to the 297 298 solubilisation of its constituents (Zhu et al., 2006). The increase in the content of crystalline cellulose in 299 the straw particles leads to enhanced thermal stability and strength of the individual biomass particles 300 (Chen et al., 2011).

301 The findings suggest that both straw samples functionalised with graphene particles, i.e., H+S-G and UN-G, have distinct XRD patterns compared to their correspondent H+S and UN straw samples. The basal 302 303 reflection peak (002) of graphene flakes at 2=26.2° is visible on the straw particles surface after the 304 functionalisation process (Lu and Ouyang, 2017; Lv et al., 2013). Moreover, (110), (200), (040), (231) and 305 (161) crystallographic planes of attapulgite particles were detected in both H+S-AT and UN-AT samples at $2\theta = 8.05^\circ$, 13.29° , 19.5° , 26.7° , and 31.02° , respectively (Tong et al., 2021). The above findings indicate 306 that straw samples were effectively functionalised by G and AT nanoparticles. The results of crystallinity 307 indices also indicated that the functionalisation of straw with AT and G induced a positive effect in terms 308 309 of increasing the crystallinity index (CI). In Table 2, the CI values increased as the surface functionalisation 310 was applied. However, the most eminent CI improvement was registered for pre-treated straw samples. 311 When compared to the counterpart sample without surface functionalisation (i.e., H+S sample), the CI 312 was enhanced by 2 % and 10 % for (H+S-G) and (H+S-AT), respectively. This improved the tensile strength of the individual biomass particles and consequently enhanced the resilience of bio-based composites 313 314 (Ghaffar et al., 2017a).

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Here→ Figure 10. X-ray diffraction patterns of untreated and H+S pre-treated wheat straw with and
 without surface functionalisations.

Here→ Table 2. Crystallinity index comparison of untreated and H+S pre-treated wheat straw particles
 with and without surface functionalisation.

320 3.4. Tensile performance improvements of PLA compressed strawboards

As shown in **Figure 11**, an investigation of straw board's tensile strength was conducted to evaluate the influence of pre-treatment and surface functionalisation processes.

323 In general, the tensile strength of bio-based composites is highly reliant on the strength of individual straw particles, straw quantity, and the quality of the interfacial bond between the straw particle and 324 PLA matrix (Ghaffar et al., 2017a; Pereira et al., 2013). In all compositions, it is evident that the tensile 325 strength of bio-based composites displayed a descending trend with the rise of straw content from 10 326 327 vol% to 50 vol%. However, the results indicated that the introduction of H+S pre-treatment appeared to 328 have a better impact on the tensile strength exhibited by the samples. The inclusion of 10, 20, 30, 40, 329 and 50 vol% pre-treated straw particles (i.e., H+S) enhanced the tensile strength by 46%, 69%, 127%, 330 66%, and 345%, respectively, compared to their corresponding bio-based composites made of (UN) straw particles. The remarkable improvement in the tensile performance of (H+S) bio-based composites 331 compared to (UN) counterparts could be associated with (i) improvements in straw's wetting properties 332 333 leading to enhanced interfacial bonding between PLA matrix and straw particles (Rakesh Kumar, Sangeeta Obrai, 2011), (ii) the cell walls expansion caused by the steaming stage (S) allowing more PLA 334 335 polymer in the liquid state to penetrate within the enlarged straw cells and thereby increasing 336 mechanical entanglement, (iii) higher crystallinity index of (H+S) straw samples, indicating that their 337 strength has improved due to the existence of more stable cellulose chains in their structure. Besides the aforementioned improvements in straw's surface functioning and PLA-straw interface, Ghaffar et al. 338 339 (2017) reported that the tensile strength of (H+S) individual strands reached 88.9 MPa, which is 35% 340 higher than that of (UN) strands (i.e., 66 MPa) (Ghaffar et al., 2017a). The tensile strength of the surface 341 functionalised composite samples was also shown to follow a similar trend. In all straw contents, H+S 342 functionalized samples (i.e., H+S-AT and H+S-G) exhibited higher or comparable overall tensile strength 343 than UN functionalised samples (i.e., UN-AT and UN-G). Moreover, the results also highlight the superior 344 impact of AT and G surface functionalisation techniques on the tensile strength of (UN) and (H+S) bio345 based composites. The most significant enhancement in tensile strength was registered in the samples 346 with the highest percentage (i.e., 50 vol %) of surface functionalised straw particles. The results indicated 347 that the AT and G surface functionalisation processes increased the tensile strength from 0.3 MPa for the (50UN) sample to 4.9 MPa and 4.3 MPa for (50UN-AT) and (50UN-G), respectively. In the case of H+S 348 349 pre-treated samples, remarkable improvements of approximately 276% and 367% were registered for (50H+S-AT) and (50H+S-G), respectively, when compared to that of (50H+S) samples. The same 350 incremental trend in tensile strength was reported by Scaffaro et al. (2020) and Zhu et al. (2019) (Scaffaro 351 352 et al., 2020; Zhu et al., 2019). As reported in previous research, surface functionalisation processes have 353 been considered to provide a crosslinking effect to enhance the interfacial bonding between straw and 354 PLA matrix (Scaffaro et al., 2020; Zhu et al., 2019). Both surface functionalising agents, i.e., graphene and attapulgite, were dispensed to improve the tensile performance of functionalised strawboards. As 355 356 shown in **Figure 12**, graphene particles on the surface of pre-treated straw effectively bridge the gap 357 between functionalised straw particles and PLA matrix, resulting in an intimate interfacial connection. It is evident that the delamination phenomenon is significantly decreased as a result of G surface 358 359 functionalisation. The effect is critical since the inclusion of a small quantity of AT and G is capable of 360 overcoming the tensile strength loss that occurs when straw particles are added, which is the most 361 prevalent drawback of bio-based composites. The developed strawboard in this study considered as a high-performance bio-based composite. The performance of the strawboard is related to production 362 method, and the matrix used and its compatibility with wheat straw. The most widely used matrices in 363 manufacturing processes are urea-formaldehyde (UF) and phenol-formaldehyde (PF). However, UF 364 offers low bonding properties between straw particles due to the hydrophobic nature of straw. It is 365 reported that strawboards bonded with UF have 6.3 MPa tensile strength on average (Wool, R. and Sun, 366 367 2011). Mo et al. (2001) has also been reported that strawboards manufactured with soy protein isolate 368 and methylene diphenyl diisocyanate adhesives provide the maximum tensile strength of 0.27 MPa and 369 0.49 MPa, respectively (Mo et al., 2003). Although several studies achieved better tensile strength than 370 this study, their preparation and manufacturing procedures (i.e., compounding and injection moulding) 371 are energy-consuming and are not feasible for producing straw boards on a large scale. A study conducted by Fan et al. (2018) investigated the performance of PLA bio-based composites reinforced by 372 373 wheat straw, which was treated with polydopamine. Their results indicated that the tensile strength of

- 374 PLA bio-based composites increased from 3.54 MPa for the samples containing untreated wheat straw
- 375 to 6.75 MPa for the samples containing polydopamine treated wheat straw (Fan et al., 2018). The
- 376 outcomes of present study indicate that the utilised pre-treatment coupled with nano functionalisation
- 377 techniques contributes to manufacturing strawboards with high-performance characteristics.
- 378 Here→ Figure 11. Tensile strength of bio-based composites with different percentage of untreated
- 379 (UN) and pre-treated (H+S) straw particles and their composites with G and AT nanoparticles.
- 380 Here→ Figure 12. SEM images of (a and b) H+S samples and (c and d) H+S-G
- 381 **3.5.** Water absorption of functionalised PLA compressed strawboards

According to acquired test results, compressed strawboards comprising 10 vol.-% and 20 vol.-% straw 382 particles and their functionalised derivatives performed the best in terms of tensile strength. Therefore, 383 384 they were selected to be assessed as appropriate compositions for physical property characterisation. 385 Water absorption (WA) test was adopted to determine the stability of PLA compressed strawboards after 386 2 and 24 hours of immersing in water. In general, the water absorption rate of bio-based composite 387 depends on several factors such as type of matrix and reinforcing biomass, temperature, humidity, and 388 biomass content. It can be seen in **Figure 13** that the inclusion of a higher straw particle content leads 389 to increased water absorption percentages which are in agreement with a previous study (Dhakal et al., 390 2007). The water absorption of all bio-based composites increased over testing time. After 2 h of 391 immersion, the results showed that water intake increased from 1.6 % and 2.3 % for 10UN and 20UN, 392 respectively, to 1.8 % and 3 % for 10H+S and 20H+S, respectively. A similar trend was also observed for 393 samples after 24 h of immersion in water. The same effect was also observed for all the compressed 394 strawboards with surface functionalised straw particles. It can be concluded that the H+S pre-treatment 395 of straw particles led to an increased water absorption of the PLA compressed strawboard. This 396 observation could be due to a reduction in the hydrophobicity of straw particles after the pre-treatment 397 as evidenced by reduced wax and silica concentration recorded from the surface chemical functional group analysis as shown in **Figure 7**. Additionally, increased surface porosity of straw particles caused by 398 399 partial hemicellulose and lignin degradation, could enable easier penetration of water molecules to the 400 compressed strawboards, as previously reported by Zeng et al. (2018) (Zheng et al., 2018). The surface functionalisation technique using both graphene nanoplatelets and attapulgite nanoclay was found to 401 be effective in decreasing strawboards water penetration. However, it has been found that samples 402

fabricated with graphene surface functionalised straws are more efficient and have slightly lower water absorption than their counterparts manufactured with attapulgite nanoclay. The results indicated that after 24 h of immersion, the water absorptions of 10UN-G, 20UN-G, 10H+S-G, and 20H+S-G are 10 %, 8 %, 20 %, and 19 % lower than the values registered for 10UN, 20UN, 10H+S, and 20H+S samples, respectively. The aforementioned results could be associated with the better compatibility of surface functionalised straws with PLA matrix, which leads to a stronger straw-matrix bonding and denser structure within the strawboards.

410 Here **Figure 13**. Water absorption of bio-composites after 2 h and 24 h of water exposure.

411 **4. Conclusion**

412 The main objective of this research was to investigate the impact of pre-treated and surface 413 functionalised straw particles as reinforcing agents in PLA compressed strawboards. Two nano functional materials (i.e., graphene nanoplatelets and attapulgite nanoclay) were employed to provide a cross-414 linking effect between straw particles and the PLA matrix. Significant alterations were observed on 415 416 wheat straw particles via characterisation tests as a result of pre-treatment and surface functionalisation 417 processes. All the bio-based composites employing graphene nanoplatelets as a surface functionalising 418 agent were shown to be the best performing composites in terms of tensile property. The 10H+S-AT and 419 10H+S-G samples exhibited maximum tensile strengths of 28 MPa and 27 MPa, respectively, which are 420 significantly higher than the 9.7 MPa recorded for the 10UN sample. Water absorption percentages of 421 1.6%, and 1.9% were registered for 10UN-G and 10H+S-G, respectively after 24h of submersion in water. 422 Employing the minimum amount of graphene nanoplatelets and attapulgite nanoclay as an emerging 423 surface functionalising (or crosslinking) agent in PLA compressed strawboards proved to be suitable for boosting straw incorporation in PLA based composites. Surface functionalised wheat straw particles 424 425 developed in this study exhibit a promising breakthrough that can be used as a benchmark in producing 426 high-performance bio-based composites.

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697 Figure 1. Schematic structure of experimental testing programme. 698 699 Figure 2. Microstructure profile of (a) graphene nanoplatelets, (b) attapulgite nanoclay and (c) particle 700 size distribution analysis of wheat straw particles

701 **Figure 3**. Schematic framework of different stages of wheat straw board preparation.

- Figure 4. SEM images of internode cross-section profile of (a) untreated (UN) and (b) pre-treated (H+S) 702 703 straw particles
- Figure 5. SEM images of (a and d) untreated (UN) straw particles and their composites with (b and e) AT 704 705 nanoclays and (c and f) graphene nanoplatelets.
- 706 Figure 6. SEM images of (a and d) pre-treated (H+S) straw particles and their composites with (b and e) 707 AT nanoclays and (c and f) graphene nanoplatelets.
- 708 Figure 7. FTIR-ATR spectrum of untreated (UN) and pre-treated (H+S) straw particles.
- 709 Figure 8. Raman spectra of untreated, and pre-treated straw particles and their composites with G and AT nanoparticles. 710
- 711 Figure 9. TGA thermogram data of untreated and H+S pre-treated wheat straw particles with and without surface functionalisation. 712
- 713 Figure 10. X-ray diffraction patterns of untreated and H+S pre-treated wheat straw with and without surface functionalisations. 714
- Figure 11. Tensile strength of bio-based composites with different percentage of untreated (UN) and 715 pre-treated (H+S) straw particles and their composites with G and AT nanoparticles. 716
- 717 Figure 12. SEM images of (a and b) H+S samples and (c and d) H+S-G.
- 718 Figure 13. Water absorption of bio-composites after 2 h and 24 h of water exposure.

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