

Development of lightweight strain hardening cementitious composite for structural retrofit and energy efficiency improvement of unreinforced masonry housings

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Abstract

The thermal, mechanical and durability properties of lightweight strain hardening cementitious composite (LSHCC) as well as the effectiveness of using LSHCC for structural retrofitting of unreinforced masonry (URM) wall is reported in this study. The proper range of water content, dosage of superplasticiser and viscosity modifying agent was explored from the survivability test of glass micro hollow bubble (3M-S15), which was much more fragile but effective in reducing the thermal conductivity of the composite than other studies. Then, the tensile properties of LSHCC with wet density of about 1,300-1,400 kg/m³ from different proportion of replacement of ordinary Portland cement (OPC) by fly ash (FA) and ground granulated blast-furnace slag (GGBS) as well as different volume fraction of polyvinyl alcohol (PVA) fibre

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were measured. The tensile ductility of LSHCC of replacement by FA was in general better than pure OPC or with GGBS blends. The tensile strength and ductility of LSHCC with 1.75% volume fraction of PVA fibre was about 3 MPa and 2-4%, respectively. The compressive strength ranged from 14 to 31 MPa. The thermal conductivity of selected LSHCC ranged from 0.34 to 0.51 W/m·K. The coefficient of water permeability of LSHCC was comparable with reference normal concrete and the ECC-M45 in the literature. The coefficient of chloride diffusivity of most LSHCC in this study was lower than the reference concrete because of the chloride binding of FA and GGBS. However, the carbonation rate of the LSHCC was generally higher. Three sets of LSHCC with similar tensile strength but different ductility were chosen for the evaluation of the effectiveness on structural retrofitting of an unreinforced masonry wall by in-plane and out-of-plane pushover analysis. The parameters of a finite element model with smeared crack material model was tuned based on the stress-strain relationship of LSHCC measured from the tensile tests in this study. There was no improvement of using LSHCC with 0.6% tensile ductility. By applying a 10 mm thick LSHCC with 2.2% and 4.4% tensile ductility on each side of an URM wall, the ductility of the retrofitted wall under in-plane loading was increased by 38% and 72%, respectively while it was increased by 164% of both kinds of LSHCC for out-of-plane loading.

Keywords:

lightweight strain hardening cementitious composites, hollow glass bubble, tensile ductility, thermal conductivity, pushover analysis, smeared crack material model

1. Introduction

Unreinforced masonry (URM) housings are vulnerable to lateral loadings such as seismic action [29, 23, 11] and wind pressure [17]. Although it is prohibited or strictly controlled to build new URM housings in many seismic regions, it is necessary to preserve the surviving/existing URM housing stock, especially some of which are historic. Confined masonry (CM) wall as an infill of reinforced concrete frame is common in low to medium height residential buildings. It can provide in-plane ductility under seismic [12, 27], however, out-of-plane collapse is still critical of old existing buildings, which were not with proper dimensioning and detailing required in modern seismic design codes [24, 53]. The collapse of confined masonry wall makes the housings no longer serviceable and may cause serious damage to adjacent structures. There are different methods to retrofit existing URM and CM housings such as base isolation [34, 49], fibre reinforced polymer (FRP) fabric, tow sheets and tapes with Kevlar and carbon fibre [22], near surface mounted (NSM) glass FRP bars [19] and carbon FRP strips [16] as well as textile reinforced mortar (TRM) with carbon fibre fabric [48, 47]. Albeit the base isolation technique is effective, it causes disturbance to the current occupant and the cost is justifiable only for valuable heritage. The drawbacks of using FRP fabric to retrofit masonry wall are the incompatibility of the polymeric matrix to common rendering/plastering materials and the low tolerance to uneven surface. The NSM FRP reinforcement retrofitting technique can resolve the stated drawbacks of FRP fabric and improve the ductility of URM and CM wall under in-plane and out-of-plane loadings, but it is labour intensive and not as effective as FRP fabric and TRM. TRM with carbon fibre fabric is a

26 promising technique, however, the stiffness of high performance man-made
27 fibre is much higher than the wall so the failure mode may be interlaminar
28 shear failure that is sudden and brittle.

29 Strain hardening cementitious composite (SHCC), which is referred as
30 engineered cementitious composite (ECC) or pseudo ductile cementitious
31 composite (PDCC) in some literature, is a family of high performance fibre
32 reinforced cementitious composite [32] based on micromechanical analytical
33 tools [42, 33, 38]. The key characteristic of SHCC is its high tensile ductil-
34 ity. The matrix can be replaced by other inorganic material such as fly ash
35 based geopolymer [45, 13, 44] and other functionalities such as lightweight
36 [54, 32, 26, 28], self-healing [3, 37, 59], low-shrinkage [13], water-repelling
37 [57] and self-sensing [1, 51]. URM wall strengthened by SHCC with hy-
38 brid steel and polyethylene fibre was proved to improve the load capacity
39 as well as ductility under quasi-static and dynamic loading [40]. Commer-
40 cial SHCC shotcrete with polyvinyl alcohol (PVA) fibre could improve both
41 the in-plane [36] and out-of-plane [35] load capacity of URM wall. Semi
42 empirical-analytical design formulas were proposed, however, those formulas,
43 which considered the tensile strength of SHCC but not the ductility, consis-
44 tently underestimated the load capacity of the strengthened URM walls from
45 experiments.

46 In addition to structural retrofit of existing URM and CM walls, it is de-
47 sirable to improve the energy efficiency by enhancing the thermal insulation
48 of the strengthening materials. In a study about the energy performance of
49 housing in south-eastern Europe, the specific heat loss through the typical
50 masonry walls is about 40% of the total heat loss through the entire building

51 envelope [41]. In another study, the energy requirement of heating of typi-
52 cal historic stone masonry housings in Italy was reduced by half and about
53 15% by adding 1.5 cm thick traditional gypsum panel (density 1000 kg/m^3
54 and thermal conductivity $0.4 \text{ W/m}\cdot\text{K}$) and plastering with glass bubble (un-
55 known thickness, density 450 kg/m^3 and thermal conductivity $0.122 \text{ W/m}\cdot\text{K}$),
56 respectively [7].

57 The wet density of conventional SHCC is about $2,000 \text{ kg/m}^3$ and the
58 thermal conductivity is about $1.2\text{-}1.5 \text{ W/m}\cdot\text{K}$ [56]. There are a few studies
59 about lightweight SHCC (LSHCC) to enhance the thermal resistance by us-
60 ing presoaked expanded perlite aggregate ($1,800\text{-}1,900 \text{ kg/m}^3$) [28, 43] and
61 fly ash cenospheres ($1,600\text{-}1,800 \text{ kg/m}^3$) [26]. [54, 32] explored four different
62 methods to reduce the density of SHCC with PVA fibre, (i) air-entrainment
63 admixture, (ii) polymeric micro-hollow bubble, (iii) natural lightweight per-
64 lite and (iv) glass micro-hollow bubble (GB) and concluded that GB demon-
65 strated superior mechanical properties than the other three approaches. [43]
66 compared the lightweight strain hardening geopolymeric composite with 2%
67 volume fraction of PVA fibre as well as expanded recycled glass (EG) and
68 microscopic hollow ceramic spheres (MS). The performance of the lightweight
69 composite is summarised in Table 1 while the properties of the two types of
70 GB (3M-S38 and 3M-S60), EG and MS are shown in Table 4. High mass frac-
71 tion of the lightweight aggregates increases the cost of LSHCC significantly.
72 Another expensive component of LSHCC is the PVA fibre.

73 In this study, LSHCC with low mass fraction ($\sim 5\%$ to total of cemen-
74 titious materials and sand filler) of GB and lower volume fraction of PVA
75 fibre ($1.5\text{-}1.75\%$) was developed. In addition to physical (density and flow

76 diameter) and mechanical (tensile strength, tensile ductility and compressive
77 strength) properties of LSHCC, other performance indicators of thermal in-
78 sulation (thermal conductivity) and durability (water permeability, chloride
79 ion diffusivity and carbonation rate) will also be reported. The applicability
80 of the developed LSHCC for structural retrofitting of masonry wall will be
81 verified based on pushover nonlinear analysis through computer simulation.

82 **2. Materials**

83 The materials used in this study for preparing the cementitious matrix of
84 LSHCC are ordinary Portland cement (OPC, CEM I 52.5), fly ash (FA, from
85 CLP Group, Hong Kong) and ground granulated blast-furnace slag (GGBS,
86 from K-Wah Construction Materials Ltd, Hong Kong). The specific gravity
87 of OPC, FA and GGBS is 3.1, 2.3 and 2.95, respectively. The results of
88 chemical composition of OPC, FA and GGBS by X-ray fluorescence (XRF)
89 spectroscopy (JEOL JSX-3201Z) are shown in Table 2. The sand used in
90 this study was Class D (between $180\ \mu\text{m}$ and $270\ \mu\text{m}$) standard silica sand
91 [8].

92 High range polycarboxylate based superplasticiser (SP, BASF Glenium
93 ACE 80, the solid content of which is about 32%) was used. Industrial
94 grade hydroxypropyl methylcellulose (HPMC) was used as viscosity modify-
95 ing agent to control segregation and bleeding of the wet mix before addition
96 of fibre.

97 Short PVA fibre (Kuraray Co. Ltd, Japan) was used to reinforce ce-
98 mentitious matrix to achieve strain hardening property of the composite.
99 It consists of 1.2% mass of oil coating on the surface to reduce the chemical

100 bonding with the cementitious matrix. The properties of PVA fibre are listed
101 in Table 3.

102 The physical, mechanical and thermal properties of the commercially
103 available GB from 3M used in this study, which is referred as S15 onwards, are
104 shown in Table 4. In order to reduce the mass fraction of LSHCC, GB with
105 larger diameter but thinner wall is used. The mean diameter of S15 is $55\ \mu\text{m}$,
106 while the diameters of the 10th percentile, 90th percentile and effective top
107 size given by the manufacturer are 25, 90 and $95\ \mu\text{m}$, respectively. The ther-
108 mal conductivity of S15 is only $0.055\ \text{W/m}\cdot\text{K}$, which is 51.2%, 72.5% and 45%
109 lower than S38, S60 and MS, respectively. However, the crush strength of
110 S15 is 2.1 MPa, which is only 7.2%, 3.0%, 4.7% of S38, S60 and MS, respec-
111 tively in the previous studies [54, 32, 43]. It is potentially to make LSHCC
112 with similar thermal insulation by using much less GB in the matrix.

113 **3. Experiment programme**

114 Since the crush strength of S15 is much lower than the lightweight aggre-
115 gates used in [54, 32, 43], it is critical to minimise the damage of S15 during
116 mixing to maintain the excellent thermal resistance. The experimental pro-
117 gramme was divided into three stages. The first stage was to determine ap-
118 propriate range of water content, dosage of SP and HPMC as well as mixing
119 time/speed of mortar, which consisted of OPC, sand and S15 only without
120 PVA fibre. The damage of S15 during mixing was indicated by the excessive
121 measured wet density compared with estimated value based on the density of
122 all ingredients and mix proportion. The second stage was to prepare LSHCC
123 samples, based on the findings about the appropriate range of water content,

124 SP and HPMC dosage from stage one, with replacement of OPC by FA and
125 GGBS and different volume fraction of PVA fibre (from 1.5% to 2%) for
126 direct tensile test to measure the tensile properties of the composite. The
127 target wet density of LSHCC is about $1,350 \text{ kg/m}^3$. The third stage was to
128 examine the compressive strength, thermal conductivity and other durability
129 parameters including water permeability, chloride diffusion and carbonation
130 rate of selected sets of LSHCC from stage 2.

131 *3.1. Stage 1: Mixing and GB survivability test*

132 Table 5 shows the 22 sets of the GB survivability test in 6 groups with
133 different water content (from 261 kg/m^3 to 353 kg/m^3) to achieve similar
134 consistency, which was indicated by flow diameter, by varying water content,
135 SP and HPMC dosage. The water content of group A was minimum with
136 maximum dosage of SP and vice versa in group F. The amount of S15 in all
137 mixes was fixed at 10% mass to cement content. However, the estimated wet
138 density varied for different groups because of different water content. For
139 group D with water content 316 kg/m^3 , the SP dosage (solid content) was
140 decreased from 0.63% to 0.1% mass to the cement content. The dosage of
141 HPMC varied from zero to 0.188% mass to the cement content.

142 All dry ingredient was dry-mixed in the Hobart Mixer HSM 20 for 7 minutes
143 at the lowest speed (speed 1). The time of wet mixing varied from 8 to 18
144 minutes. The mixing speed of wet mixing was at the lowest speed except A2
145 and A3, which was set at medium speed (speed 2). After the wet mixing,
146 the fresh mix was poured into 100 mm cubic steel mould and compacted on a
147 vibrating table. After wiping any excess outside the mould, it was weighted
148 in an electronic balance with $\pm 5 \text{ g}$ accuracy. The reported plastic density

149 was the average of three samples from the same batch of mix.

150 The consistency of the fresh mortar was measured according to modified
151 flow table test from BS EN 1015-3:1999 [9]. The modification was to skip
152 the vibration after compaction and mould raising. The flow diameter was
153 measured 2 times of each test with accuracy up to ± 1 mm. The diameter
154 reported was the range of flow diameters from 3 repeated tests and rounded
155 to 5 mm.

156 *3.2. Stage 2: Mixing, curing and testing of LSHCC*

157 The specimens of LSHCC for direct tensile test were divided into 3 groups
158 (Table 6). All mixes consisted of about 30% volume fraction of S15. The
159 targeted wet density was about $1,350 \text{ kg/m}^3$. Group 1 (GI) consisted of
160 OPC, S15 and sand. Group 2 (GII) consisted of OPC, FA, S15 and sand.
161 Group 3 (GIII) consisted of OPC, FA, GGBS, S15 and sand. OPC, FA,
162 GGBS, S15, sand, HPMC, if applicable, was mixed in Hobart Mixer HSM
163 20 at the lowest speed for 7 minutes. SP was mixed with water and the
164 mixture was then added to the dry mix and mixed at the lowest speed for
165 another 9 minutes. PVA fibre was added and mixed for further 5 minutes at
166 the lowest speed to form LSHCC. The wet density and consistency of fresh
167 LSHCC was measured by the same method in section 3.1.

168 In addition to fresh properties of LSHCC, tensile test samples were pre-
169 pared. The dimensions of the plate-shape specimen for direct tensile test of
170 LSHCC were $350 \text{ mm} \times 50 \text{ mm} \times 15 \text{ mm}$ (Figure 1). 3 specimens were prepared
171 of each mix. The specimens were covered by cling wrap at room temperature
172 after casting. Then, they were demoulded and cured at 25°C and 98% rel-
173 ative humidity for 27 days. After 28 days from casting, the specimens were

174 air dried for 1 day. In order to strengthen the region of the specimen under
175 high gripping force during test and prevent cracks form in that region, a layer
176 of carbon fibre reinforced polymer (CFRP) composite (100 mm×50 mm) was
177 glued by epoxy on both ends of one of the surface and a 1.2 mm thick alu-
178 minium sheet (70 mm×50 mm) was attached on top of each CFRP sheet.
179 After the resin of CFRP was cured for 24 hours at room temperature, the
180 same procedure was repeated to the other surface of the specimen. The ten-
181 sile test was performed between the 31st and 35th day from casting. The
182 tensile test was carried in MTS 810. During the test, a pair of liner variable
183 differential transformers (LVDTs) were mounted at the edge of the surface
184 of carbon fibre layer. At the other end of the LVDT, a pair of fixed plates
185 with an adjustable screw were glued on the side of the specimen by 2-part
186 araldite epoxy adhesive. The loading rate was set at 0.1 mm/min.

187 *3.3. Stage 3: Compressive strength, thermal and durability tests of LSHCC*

188 Compressive strength, thermal conductivity, water permeability, chloride
189 diffusivity and carbonation rate of LSHCC was tested only on selected mixes
190 but from different batch of mixing followed the identical mixing procedure
191 described before. Three 100 mm cubic samples were prepared for compression
192 test and they were covered by cling wrap for 24 hours in the room temper-
193 ature (about $23\pm 1^\circ\text{C}$) of laboratory after casting. Then, the cubic samples
194 were demoulded and cured at 25°C and 98% R.H. for further 27 days. The
195 cubic samples were tested in ELE automatic compression machine with load-
196 control at 3 kN/s loading rate. The reported compressive strength was the
197 average of three samples from the same batch.

198 The coefficient of thermal conductivity was measured by hot-wired method

199 (QTM-500, Kyoto Electronic). A 100 mm cubic sample was prepared followed
200 with the same curing procedures described for compressive strength test. Af-
201 ter curing, the sample was oven-dried at 115°C for 24 hours and then cooled
202 down to room temperature for another 24 hours. The reported coefficient
203 of thermal conductivity was the average of three measurements from three
204 different faces of the same cubic sample.

205 The coefficient of water permeability was measured by modified falling
206 head test [31]. The dimensions of the sample were 130 mm×50 mm×15 mm.
207 The samples were cured in the room temperature in laboratory for 28 days
208 before the test. The water reservoir was made of poly(methyl methacrylate)
209 (PMMA) and all edges were sealed by epoxy (Figure 2). The internal di-
210 mensions of the water reservoir were 120 mm×40 mm. The top and bottom
211 chambers were filled by water and the measurement was started after 2 weeks
212 so that the sample was saturated. The top of the standpipe was covered by
213 cling wrap to minimise water loss during the test. The water head was mea-
214 sured twice a week for 4 weeks. The coefficient of water permeability (k)
215 can be estimated from the linear fit of the plot of natural logarithm of the
216 ratio of initial to final water head versus time according to the Darcy's law
217 in Eq. (1).

$$k = \frac{a \cdot d}{A \cdot t_f} \ln \frac{h_i}{h_f} \quad (1)$$

218 where a , A , d , t_f , h_i and h_f are the area of the standpipe, the area of
219 the reservoir, thickness of the specimen, time taken, initial water head and
220 final water head, respectively. The reported coefficient of water permeability
221 is the average of three specimens from the same batch of mix.

222 The coefficient of chloride diffusivity was estimated by the hybrid of non-
 223 steady state migration test and colorimetric method [5, 6]. Cylindrical spec-
 224 imens with dimensions of 100 mm diameter and 50 mm thick were prepared
 225 for non-steady state migration test. The specimens were cured at the room
 226 temperature of laboratory environment for 28 days. The circumferential sur-
 227 face was sealed by epoxy and vacuum saturated in water. The upstream
 228 and downstream reservoir was filled by 3% mass of sodium chloride solu-
 229 tion and 0.1 N sodium hydroxide solution, respectively. A #30 copper mesh
 230 was attached on each flat surface of the cylindrical sample and they were
 231 connected with 30 V direct current (Figure 3). After 48 hours, the specimen
 232 was split into 2 halves to reveal the fresh surface. The depth of chloride
 233 penetration (x_d) was measured from the colour change by spraying 0.1 N
 234 silver nitrite (AgNO_3) aqueous solution, at which the free chloride amount
 235 ($30.5 \text{ mol/m}^3 = 0.03 \text{ N}$) at the colour-change boundary was similar to the chlo-
 236 ride threshold value of corrosion [30], on the fresh surface. The coefficient of
 237 non-steady state migration chloride diffusion (D_{nss}) is given by Eq. (2).

$$D_{nss} = \frac{1}{a \cdot t} \left[x_d - 2\sqrt{\frac{x_d}{a}} \varepsilon \right] \quad (2)$$

where t is the duration in second and

$$a = \frac{|Z| \cdot F \cdot E \cdot t}{R \cdot T \cdot d} \quad (3)$$

$$\varepsilon = \text{erf}^{-1} \left(1 - \frac{2C_d}{C_0} \right) \quad (4)$$

238 where R is the gas constant ($8.314 \text{ J/mol}\cdot\text{K}$), T is the absolute temper-
 239 ature, E is the potential difference between anode and cathode, d is the
 240 thickness of the specimen, Z is the valence of ion, F is Faraday constant

241 ($96485 \text{ C}\cdot\text{mol}^{-1}$), $\text{erf}^{-1}()$ is the inverse error function, C_0 (0.512 N) and C_d
242 (0.03 N) is the molar concentration of chloride ion at the upstream surface
243 and the colour-change boundary, respectively.

244 100 mm cubic specimens were prepared for accelerated carbonation test.
245 The specimens were cured at room temperature in laboratory for 28 days.
246 Then, they were dried in an oven at 60°C for 3 days. After they were
247 air-cooled in laboratory to room temperature, they were sealed by paraf-
248 fin. The specimens were put in a carbonation chamber (CABR-HTX12)
249 with $5\pm 0.2\%$ CO_2 at $20\pm 1.5^\circ\text{C}$ and $70\pm 5\%$ relative humidity for 927 hours.
250 After 927 hours, the specimens were cut by wedged compression. The car-
251 bonation depth was determined by colorimetric method by using solution of
252 phenolphthalein indicator prepared according to [10].

253 4. Results and discussions

254 4.1. GB survivability test

255 One important indicator of the survival rate of S15 is the wet density.
256 If significant portion of S15 was broken during the mixing process, the wet
257 density measured was much higher than the targeted density estimated from
258 the specific gravity of the raw materials and mix proportion. The flow di-
259 ameter and percentage error of the measured from the targeted wet density
260 is shown in Figure 4. Although the flow diameters of groups A, B, C and
261 D1-D4 were in similar range about 300 mm, the breakage of S15 in group
262 A is much higher than groups C and D1-D4. The fluidity of group A was
263 mainly by high dosage of SP with low water content. On the contrary, the
264 water content was higher in group D with lower dosage of SP. The reason

265 was that the fluidity contributed by SP mainly by the shear stress induced
266 during mixing and it might break S15 before the SP became effective. When
267 the fluidity was from higher water content, the effect was much faster than
268 SP and it avoided excessive shear stress, which might break the S15, at the
269 initial stage of wet mix. By comparing D2 and D5, when the dosage of SP
270 is reduced by half, the fluidity of the mix dropped significantly. When the
271 dosage of SP was less than 0.4% (D3-D11, E1 and F1), the breakage of S15
272 was significant (more than 10%) even through the ultimate fluidity was sim-
273 ilar. When the dosage of SP was less than 0.2%, for some cases, the effect of
274 SP could be activated only after much longer duration of mixing (D8-D11).
275 For the given water content and SP dosage, the breakage of S15 increased
276 with increased dosage of HPMC. The addition of HPMC increased the vis-
277 cosity of fresh mix. By comparing D5-D7, with 316 kg/m^3 water content and
278 0.2% SP dosage, the flow diameter increased from 220 mm to 350 mm when
279 the dosage of HPMC decreased from 0.15% to 0%. When the dosage of SP
280 increased with the given water content, higher HPMC dosage could be used
281 to achieve similar flow diameter (D1-D4 and D6). In summary, the general
282 guidelines for the mix design of using S15, which is much more fragile com-
283 pared to the lightweight aggregates used in other literature, are that (i) the
284 water content is about 300 kg/m^3 , (ii) SP content is at least 0.4% and (iii)
285 the HPMC content is about 0.1%, to achieve desirable survival rate of S15
286 after mixing.

287 *4.2. Tensile test of LSHCC*

288 The stress-strain curves of the tensile test of GI (OPC-sand blend), GII
289 (OPC-FA-sand blend) and GIII (OPC-FA-GGBS-sand blend) are shown in

290 Figures 5, 6 and 7, respectively. The first crack strength, ultimate tensile
291 strength and tensile ductility is shown in Figure 8. The tensile ductility is
292 defined as the tensile strain corresponding to the ultimate tensile strength.
293 The error bars in Figure 8 represent the 90% confidence interval of the first
294 crack and ultimate tensile strength based on the three experimental results
295 by $\mu \pm t_{0.05,2} \cdot \sigma / \sqrt{2} = \mu \pm 2.920\sigma / \sqrt{2}$, where $t_{0.05,2}$ is the upper 5 percentile
296 of the t-distribution with 2 degrees of freedom, μ and σ are the mean and
297 standard deviation of the three samples, respectively. The three strokes (top,
298 bottom and middle) of the uniform bars in Figure 8 show the tensile ductility
299 of the three tensile tests. The ductility is classified as low, medium and high
300 corresponding to GI-2:3 (0.69%), GI-4:1 (2.15%) and GI-2:1 (4.70%), which
301 will be used to demonstrate the effectiveness of using LSHCC with different
302 tensile ductility to retrofit unreinforced masonry wall in section 5.

303 In GI, GI-1, GI-2 and GI-4, with 2% volume fraction of fibre, exhibited
304 low to high tensile ductility. The aggregates (sand + GB) to binder ratios of
305 GI-1, GI-2 and GI-4 were 0.057, 0.225 and 0.58, respectively. The first crack
306 strength of GI-1 (2.64 MPa) and GI-4 (2.63 MPa) was similar while that of
307 GI-2 (1.99 MPa) was significantly lower. It might be because the air content
308 of GI-2 was higher by the high negative percentage error of the measured
309 wet density relative to the estimated one. The ultimate tensile strength of
310 GI-1 was higher than GI-2 and GI-4. It might be because of the better bond
311 strength at the fibre-matrix interface of higher binder content of GI-1. GI-3
312 and GI-5 with fibre volume fraction of 1.75% did not exhibit strain hardening
313 behaviour but as conventional fibre reinforced concrete. The flow diameters
314 of GI-3 and GI-5 were in the range of 120-130 mm which were smaller than

315 GI-1, GI-2 and GI-4 (between 135 mm and 170 mm). The percentage error
316 of the measured wet density relative to the estimated one of GI-3 and GI-5
317 was significantly higher than GI-1, GI-2 and GI-4. It indicated that GB was
318 damaged during mixing and it is detrimental to the multiple-crack formation.

319 The fibre volume fraction of GII was 1.75% except GII-7 (1.50%). The
320 flow diameters were generally higher (180 mm – 200 mm) than GI because
321 of the spherical morphology of FA particle except GII-1 (160 mm) and GII-5
322 (140 mm), of which the FA to OPC ratio was lower and the tensile ductility
323 was significantly lower than the other five. With the higher flow diameter,
324 the variation of stress-strain relationship of the same mix proportion was less.
325 For GII-1 and GII-5, the flow diameter was the smallest and the consistency
326 of the stress-strain relationship was the lowest. The first crack strength
327 of GII-2 and GII-3 was lower than the other five because the air content
328 was higher deduced from the negative percentage error of the measured wet
329 density relative to the estimated one. The main difference between GII-
330 6 (1.75% vol) and GII-7 (1.50%) was the fibre volume fraction with similar
331 flow diameter. Although the first crack strength and ultimate tensile strength
332 of GII-7 was about 12% and 14% lower than GII-6, respectively, the tensile
333 ductility of them were similar (3.5% for GII-6 and 3.31% for GII-7). That
334 means the tensile ductility of LSHCC with dry density about 1,250 kg/m³
335 can be maintained at medium to high range.

336 In GIII, the tensile ductility was in the range of low to high ductility ex-
337 cept GIII-4, which did not exhibit strain hardening behaviour. GIII-4 did not
338 contain any FA. The flow diameter was only 130 mm and the percentage error
339 of the measured wet density was significantly higher than GIII-1, GIII-2 and

340 GIII-3. The flow diameter of GIII-1 was the highest and the tensile ductility
341 was the highest. Although the fibre volume fraction of GIII-3 was 2.00%
342 compared to 1.75% of GIII-2, the tensile ductility was similar. Although the
343 GGBS content of GIII-2 was higher than GIII-3, the first crack strength and
344 ultimate tensile strength of GIII-2 was higher than GIII-3. Similar to the
345 observation of GI and GII, the first crack strengths of GIII-1 (2.37 MPa) and
346 GIII-3 (2.48 MPa) were lower than GIII-2 (2.71 MPa) and GIII-4 (3.19 MPa)
347 because of the higher air content indicated by the negative percentage error
348 of the measured wet density relative to estimated one.

349 *4.3. Compressive strength, thermal conductivity and durability parameters*

350 The compressive strength, thermal conductivity, water permeability, chlo-
351 ride diffusivity and carbonation rate were measured for the selected set of
352 samples. For comparison and cross-reference to other literature, those engi-
353 neering properties of reference concrete samples with unknown mix formula-
354 tion for precast reinforced concrete building façade (C35/45), provided by
355 a local concrete producer in Hong Kong, were also measured and named Ref
356 in Figure 9.

357 Figure 9a shows the density, compressive strength and coefficient of ther-
358 mal conductivity of the selected groups of samples. The compressive strength
359 of the reference concrete sample (C35/45) was 54 MPa. Since the aggre-
360 gates (sand + GB) to binder ratio of GI-4 was 0.58 compared to GI-2 of
361 0.225, the compressive strength of GI-4 (13.6 MPa) was 41% lower than GI-
362 2 (23.2 MPa). The compressive strength of GII-3, GII-4, which consisted
363 of high fraction of FA (OPC-to-FA ratio = 1:4), was lower compared with
364 other samples as expected because of the low reactivity of FA. However, with

365 OPC-to-FA ratio 2:1 and higher density, the compressive strength of GII-5
366 was 31.1 MPa. The compressive strength of GIII-2 (25.9 MPa) and GIII-3
367 (24.7 MPa) was comparable with GI-2. Since GIII-1 consisted of 60% of FA
368 in the binder and the density is lower, the compressive strength of GIII-1
369 (19.8 MPa) was about 22% lower than GIII-2 and GIII-3. The compressive
370 strength of LSHCC with the dry density of about 1,200-1,300 kg/m³ in this
371 study was lower than the values reported in [32] that the compressive strength
372 was 41.2 MPa and 21.8 MPa for density of 1,460 kg/m³ and 930 kg/m³, re-
373 spectively. However, the GB used in [32] was S60 and S38 for the low and
374 ultra-low density LSHCC while S15, the thermal conductivity of which was
375 about half and 27% of S60 and S38, respectively, was used in this study. It
376 is more effective to use S15 for thermal insulation application. 20% mass of
377 S60 and 50% mass of S38 to cement was used in [32] while there was about
378 5-6% to total binder employed in this study. Hence, the material cost of the
379 LSHCC in this study is significantly lower. Since the crush strength of S15 is
380 about 7.6% and 3.0% of S60 and S38, respectively, the compressive strength
381 is expected to be lower than LSHCC with S60 or S38.

382 The coefficient of thermal conductivity (λ) of the reference concrete was
383 2.08 W/m·K which was at least about four times higher than the selected
384 set of LSHCC in this study. In GI, the dry density of GI-2 and GI-4 was
385 the same, but the λ -value of GI-4 (0.42 W/m·K) was about 21% lower than
386 GI-2 (0.53 W/m·K) while the sand content of GI-4 was about three times of
387 GI-2. It can be explained by the high porosity indicated by low compressive
388 strength [55] of GI-4 (13.6 MPa) compared with GI-2 (23.2 MPa). In GII,
389 the compressive strength of GII-3 and GII-4 was similar, but the density

390 of GII-3 (1,119 kg/m³) is lower than GII-4 (1,277 kg/m³). It indicated high
391 porosity of the matrix of GII-4, so the λ -value of GII-4 was lower than GII-
392 3. For GII-5, the density and compressive strength was higher than GI-2
393 but the λ -value of GII-5 was smaller. The λ -value decreased with increased
394 replacement of OPC by FA up to 30% of replacement [14, 15, 52]. For GIII,
395 the λ -value decreased with density. By comparing GIII-3 and GI-2, both
396 the compressive strength and density of GIII-3 was higher than GI-2 but the
397 λ -value of GIII-3 was lower than GI-2. It was because of the replacement
398 OPC by GGBS [14, 52].

399 Figure 9b shows the results of the test of water permeability. The plot is
400 in semi-log of the y-axis. The height of the bars represented the mean value
401 and only the upper bound of the error bar for the 90% confidence interval
402 is shown. The coefficient of water permeability of ECC-M45 report in [31]
403 is shown in the dash line in Figure 9b. All of them were comparable with
404 the reference normal concrete and ECC-M45 except GII-3 and GII-4. GII-3
405 and GII-4 contained high proportion of FA (FA-to-OPC ratio = 4:1) and the
406 compressive strength was low, which indicated high porosity of the matrix.

407 The results of the coefficient of chloride diffusivity (D_{nss}) are shown in
408 Figure 9c. The value of D_{nss} of the reference concrete was 1.59×10^{-11} m²/s.
409 In GI, D_{nss} of GI-2 was lower than the reference concrete as expected be-
410 cause there was no transition zone in GI-2. While the coefficient of water
411 permeability of GI-4 was similar to GI-2, D_{nss} of GI-4 was much higher than
412 GI-2 because the cement content of GI-2 was much higher than GI-4 and
413 there might be chloride binding by C₃A in OPC [58]. In GII, although the
414 water permeability of GII-3 and GII-4 was higher than the reference concrete,

415 D_{nss} was lower than the reference concrete because of chloride binding of FA
416 [58]. When the water permeability of GII-5 was similar with the reference
417 concrete, D_{nss} was only 2.6% of the reference concrete. In GIII, the water
418 permeability of GIII-1 and GIII-2 was similar to the reference concrete, D_{nss}
419 was much lower because of chloride binding of GGBS [39]. When the water
420 permeability of GIII-3 was lower, D_{nss} was further reduced.

421 Carbonation depends on the porosity, internal moisture content and the
422 availability of Portlandite. Figure 9d shows the results of carbonation rate.
423 The carbonation rate of the reference concrete was 3.8 mm/month^{1/2}. In
424 general, the replacement of OPC by FA and GGBS increases the carbonation
425 rate for the same strength grade because the pozzolanic reaction of FA and
426 GGBS consumes Portlandite although the pore structure is refined [4, 18, 25].
427 However, the compressive strength of different mixes was different so the
428 carbonation rate was not comparable directly. When the carbonation rate
429 and compressive strength of the mixes with FA and GGBS (GII and GIII) was
430 plotted (Figure 10, the relationship follows a linear line ($R^2=0.91$)). However,
431 if the linear relationship is used for GI, it overestimates the carbonation rate
432 from compressive strength. It is because there is no pozzolanic reaction in
433 GI so more Portlandite for carbonation reaction.

434 **5. Pushover analysis of unreinforced masonry wall strengthened by** 435 **LSHCC**

436 Pushover analysis was conducted for masonry walls with and without
437 LSHCC in order to investigate the effects of the LSHCC on the in-plane
438 and out-of-plane lateral force resisting capacities of a unreinforced masonry

439 wall. Three sets of LSHCC with different ductility but similar strength were
440 selected from the experimental results of the tensile test in section 4.2 for the
441 analysis corresponding to high (GI-2:1), medium (GI-4:1) and low (GI-2:3)
442 tensile ductility. The meaning of GI-2:1 was the stress-strain relationship of
443 the first curve in GI-2 (OPC-sand blend). The dimensions of a typical low
444 height-to-length ratio masonry wall in the analysis were $6\text{ m}\times 3\text{ m}\times 0.23\text{ m}$.
445 The thickness of LSHCC was 10 mm thick applied on both sides of the wall.

446 *5.1. Finite Element Model and Validation*

447 Pushover analysis was conducted through a three-dimensional model in
448 general finite element software ANSYS (Figure 11). The wall was fixed at the
449 base and subjected to uniformly distributed load at the top. At each step,
450 incremental displacement was applied at the top of wall along and perpen-
451 dicular to the wall surface for in-plane and out-of-plane pushover analysis,
452 respectively.

453 The masonry wall was modelled as 3D solid elements with unreinforced
454 smeared crack material models combined with multilinear isotropic plasticity
455 as proposed by [2]. The mechanical properties of the masonry were referred to
456 the test results in [20] and they are shown in Table 1. The parameters of the
457 unreinforced masonry wall model were calibrated from the in-plane pushover
458 experiment in [21] and the calibrated results are shown in Figure 12a.

459 The LSHCC was modelled as 3D solid elements with reinforced smeared
460 crack material models [50, 2]. The first crack strength, ultimate tensile
461 strength and tensile ductility of the selected experimental data was retrieved
462 from section 4.2. Other properties such as tensile stress-strain inputs and
463 volumetric ratio of the reinforcement were calibrated to the stress-strain re-

464 lationship in section 4.2. The calibrated/simplified constitutive relations of
465 LSHCC are shown in Figure 12b.

466 *5.2. Analytical Results*

467 Figure 13 shows the pushover response (base shear versus drift at the top
468 of the wall) for (a) in-plane loading and (b) out-of-plane loading. Several
469 critical states are indicated as markers in Figure 13 including the first crack
470 in masonry, crushing in masonry, crushing in LSHCC and tensile fracture
471 in LSHCC elements. For in-plane loading case, the LSHCC significantly in-
472 creased the strength of the wall from 43% to 76% depending on the ductility
473 of the LSHCC as seen in Figure 13a. The LSHCC did not change the drift
474 capacity where the wall started crack due to the strain compatibility but it
475 influenced the post-cracking behaviour of the wall. The low ductility LSHCC
476 had a lower overall ductility compared to the wall without LSHCC because
477 the failure of the wall with low ductility LSHCC was controlled by the rup-
478 ture of the fibre. The medium and high ductility LSHCC increased overall
479 ductility of the wall. The failures of the wall with medium and high duc-
480 tility LSHCC were initiated from the crushing of the masonry and LSHCC
481 elements respectively. The masonry element started to crush earlier in the
482 wall with medium ductility LSHCC than high ductility LSHCC and it might
483 be because the medium ductility LSHCC was slightly stronger than the high
484 ductility LSHCC (refer to Figure 12b). The stronger LSHCC could provide
485 larger confinement to the masonry and increase the force transferred through
486 the wall, which in turn made the masonry element to crush earlier.

487 For the out-of-plane loading case, the LSHCC also showed significant
488 increase in the strength of the wall as seen in Figure 13b. The medium and

489 high ductility LSHCC significantly increased the overall ductility of the wall.
490 The overall ductility of the wall increased as the increase of the ductility in the
491 LSHCC material because the failure of the wall was controlled by the tensile
492 fracture of LSHCC. Compared to the in-plane loading case, the LSHCC was
493 more effective to increase the overall ductility for the out-of-plane loading.
494 The higher ductility achieved in the out-of-plane loading was possibly because
495 the response of the wall was controlled by flexural deformation in the out-
496 of-plane loading while it was controlled by shear deformation in the in-plane
497 loading. Figure 14 shows the crack pattern in the masonry wall element
498 at 0.3% drift for the wall without LSHCC and with the medium ductility
499 LSHCC under the in-plane loading. The wall with LSHCC had more cracks
500 than the wall without LSHCC because the confinement of the LSHCC allowed
501 the masonry wall to transfer more forces and resulted in more cracks in the
502 masonry wall.

503 Figure 15a shows the relationship between plastic strain in the LSHCC
504 and the drift at top of the wall. The plastic strain is normalized by the
505 ultimate strain which is defined as the strain at peak strength of LSHCC
506 (refer to Figure 12b). The fibre in the low ductility LSHCC reached its
507 ultimate strain in both in-plane (at 0.26% drift) and out-of-plane (at 1.2%
508 drift) loadings which indicated its insufficient ductility for strengthening the
509 masonry wall. The fibre in the medium ductility LSHCC just reached the
510 ultimate strain at a fairly large drift (3.7%) for the out-of-plane loading while
511 it did not reach the ultimate strain under the in-plane loading. The fibre in
512 the high ductility LSHCC did not reach the ultimate strain under both in-
513 plane and out-of-plane loadings.

514 The rigid and continuous model assumption between the masonry wall
515 and LSHCC was examined by the bond stress at the LSHCC/masonry in-
516 terface. Figure 15b shows the bond stress between the wall and LSHCC
517 interface. As seen, the bond stress was higher in the out-of-plane loading
518 case than that in the in-plane loading case due to higher drift occurred in
519 the out-of-plane loading. The maximum bond stress was much lower than
520 the bond strength (0.24 MPa) estimated from the tests in fibre reinforced
521 cementitious matrix (FRCM) composite reported in literature [46].

522 **6. Conclusions**

523 LSHCC based on S15 GB with dry density about $1,350 \text{ kg/m}^3$ has been
524 developed that can achieve 2-4% tensile strain. The proposed survivability
525 test of mortar can provide a guideline of water content as well as the dosage
526 of superplasticiser and viscosity modifying agent. The increase of water con-
527 tent is more beneficial for the survivability of hollow glass bubble than SP
528 because the initial excessive shear stress during mixing may damage the GB.
529 The tensile ductility of OPC-FA-sand blend and OPC-FA-GGBS blend was
530 generally better than the OPC-sand blend while the OPC-FA-blend was the
531 best in this study. The compressive strength depended on the density as well
532 as the porosity in the matrix. The thermal conductivity of LSHCC is about
533 25% of normal structural concrete. The coefficient of water permeability of
534 LSHCC is comparable to normal concrete. The coefficient of chloride diffusiv-
535 ity is commonly lower than normal concrete because of the chloride binding
536 of FA and GGBS. However, the carbonation rate of LSHCC is commonly
537 higher than normal concrete.

538 The experimental stress-strain relationship of LSHCC under tensile was
539 used for pushover analysis of a unreinforced masonry wall. From the pushover
540 analysis results, the LSHCC can increase the strength and ductility of the
541 masonry under both in-plane and out-of-plane loadings by providing the con-
542 finement and allowing more forces transferring through the masonry element.
543 To ensure an efficient retrofit for the masonry, the LSHCC needs to have a
544 sufficient ductility. The LSHCC is more effective on increasing the overall
545 ductility of the wall for the out-of-plane loading due to the flexural controlled
546 deformation in this direction.

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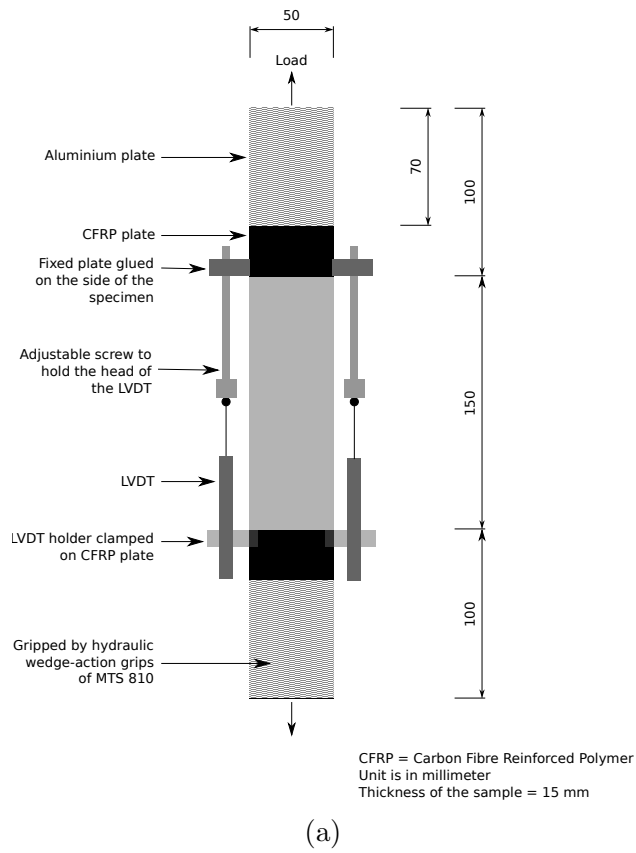
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(b)

Figure 1: (a) Schematic diagram of the direct tensile test of SHCC. (b) Tensile test configuration.

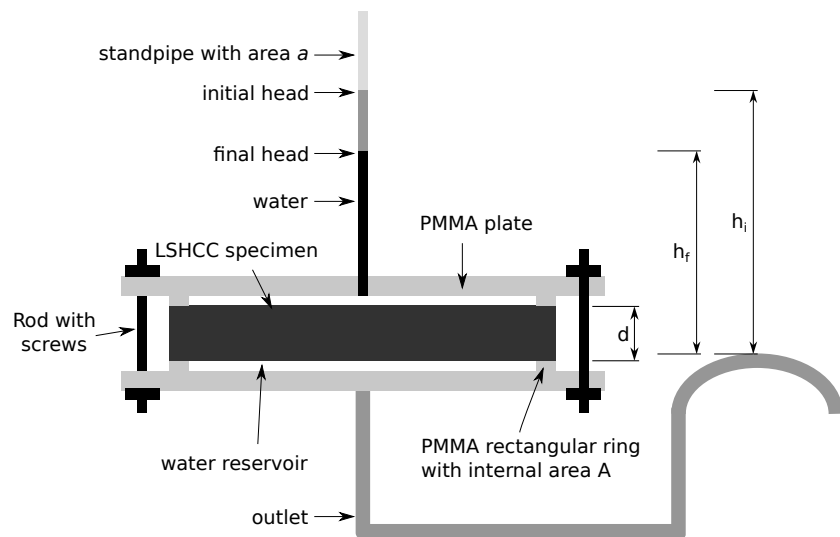


Figure 2: Schematic diagram of the falling head test.

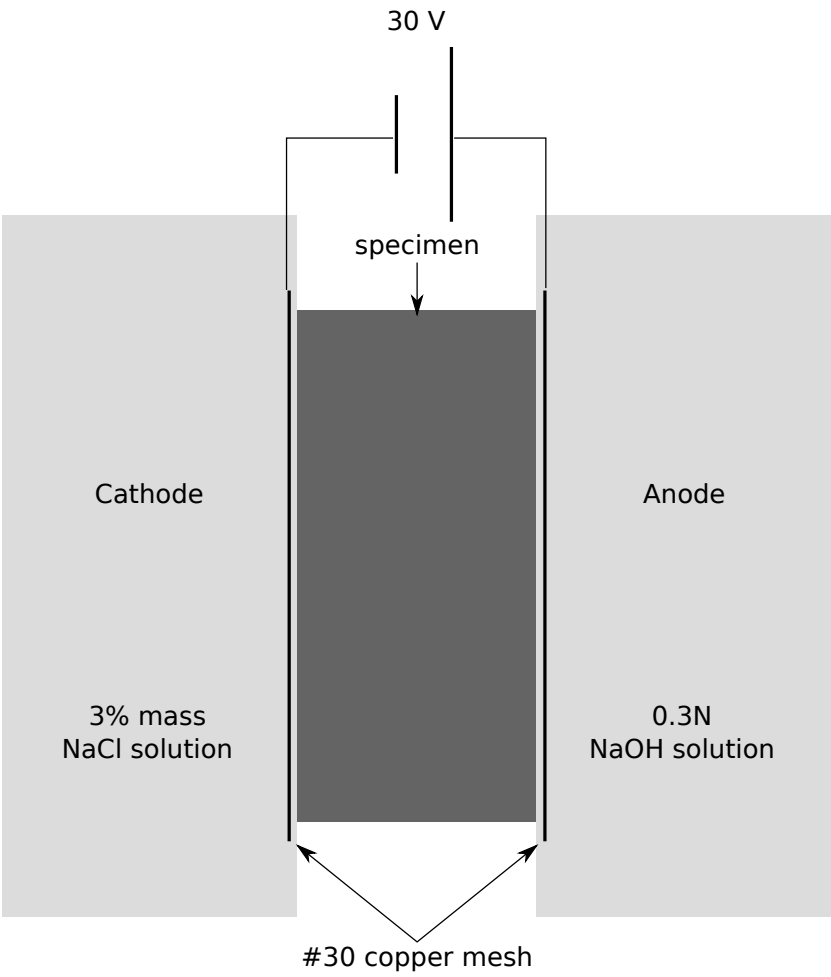


Figure 3: Schematic diagram of rapid chloride diffusion test.

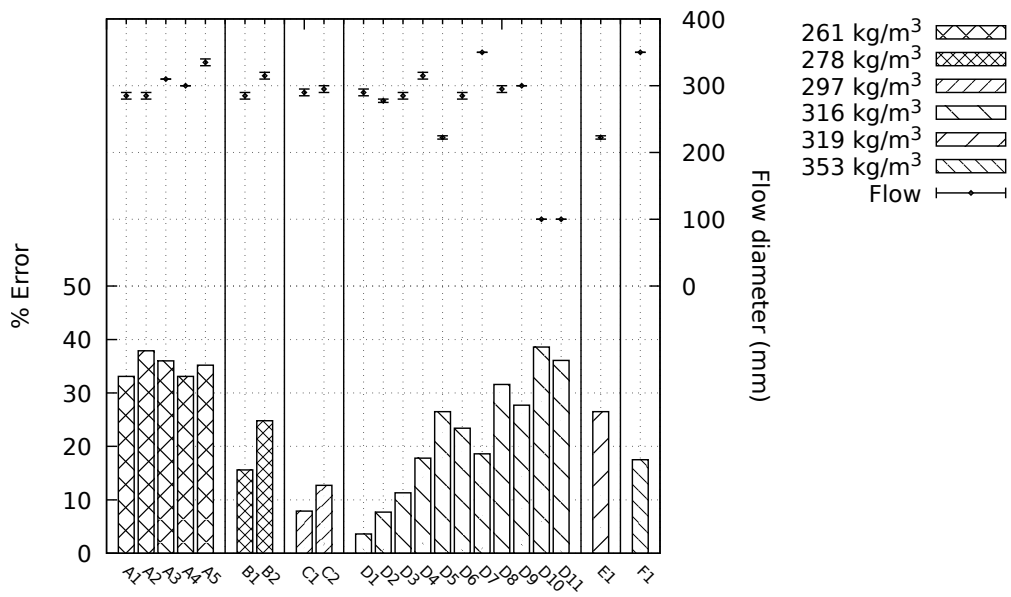


Figure 4: Deviation of wet density and flow diameter results of mortar test.

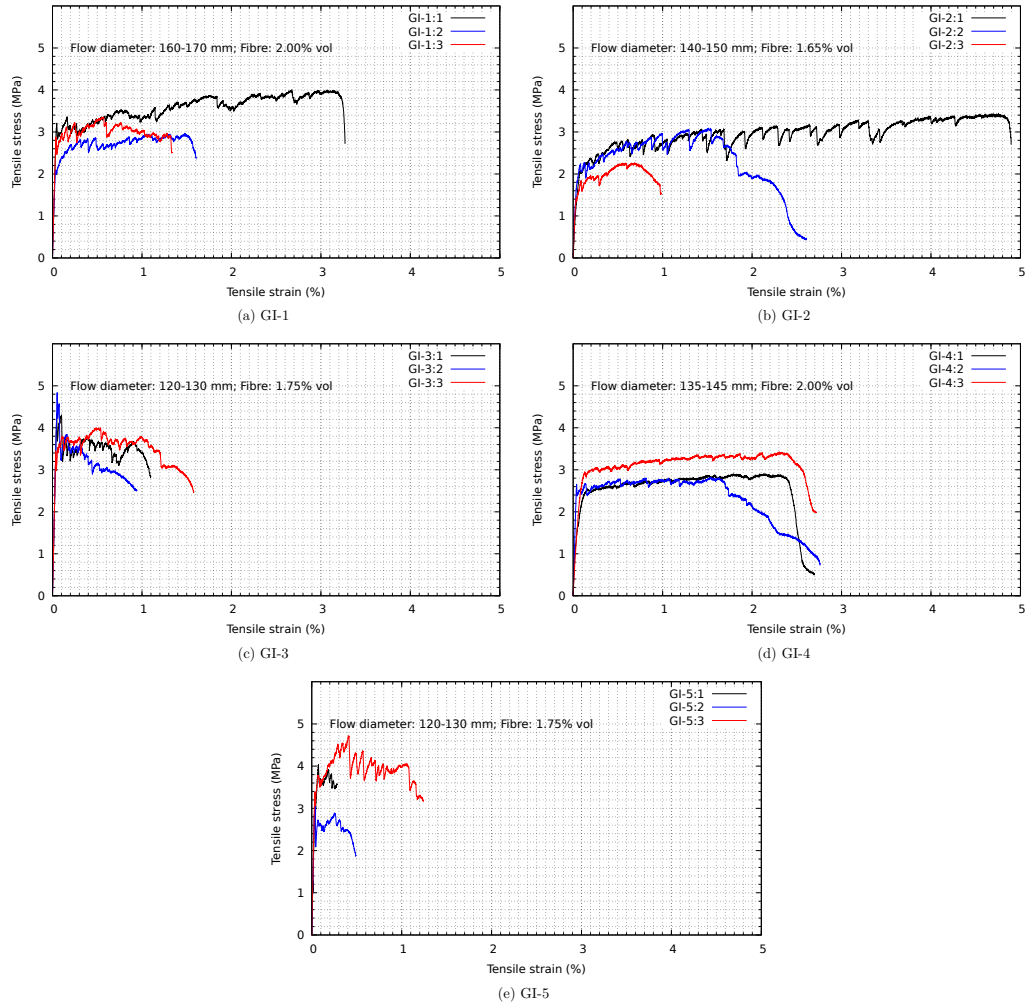


Figure 5: Results of tensile test of the lightweight composite of GI.

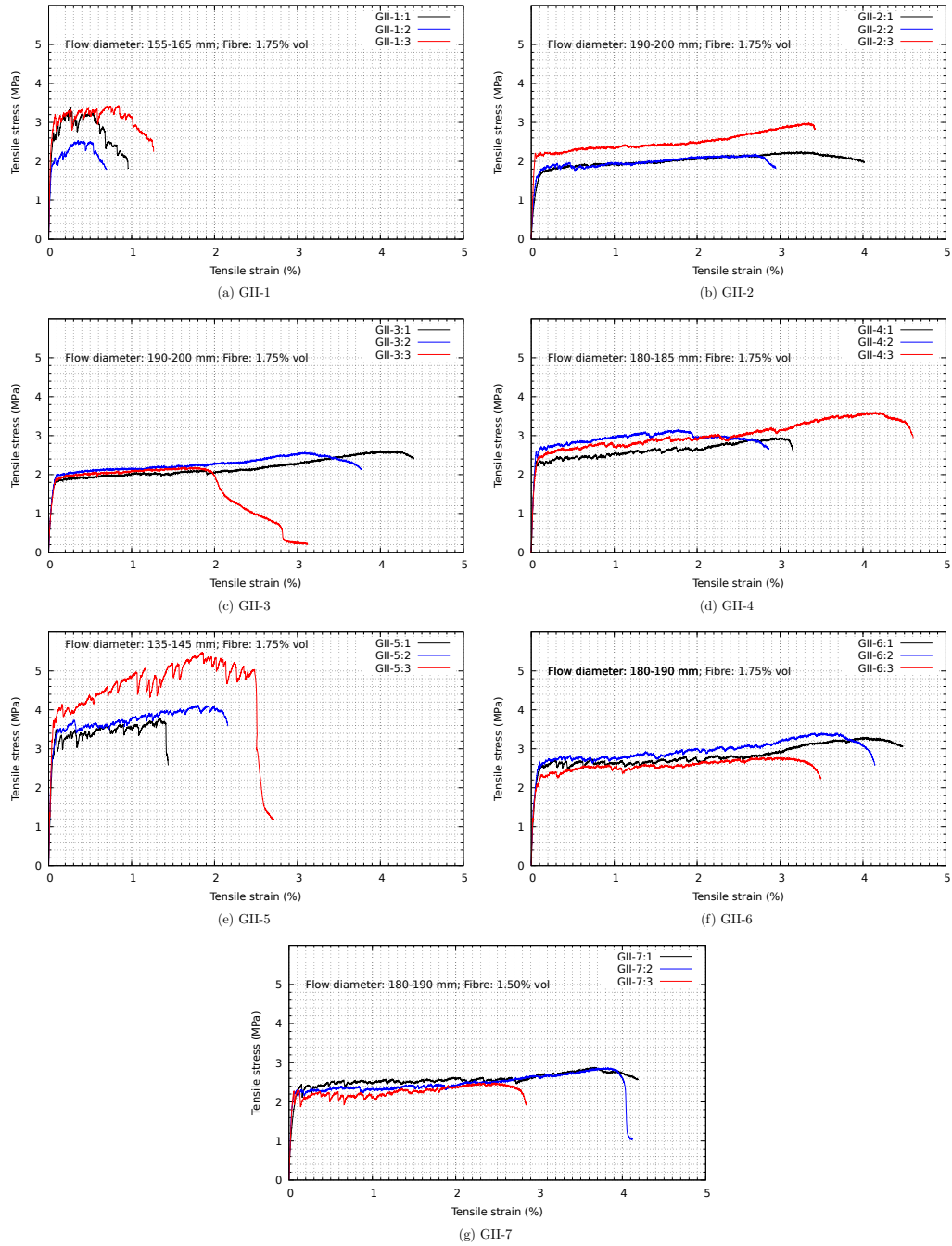


Figure 6: Results of tensile test of the lightweight composite of GII.

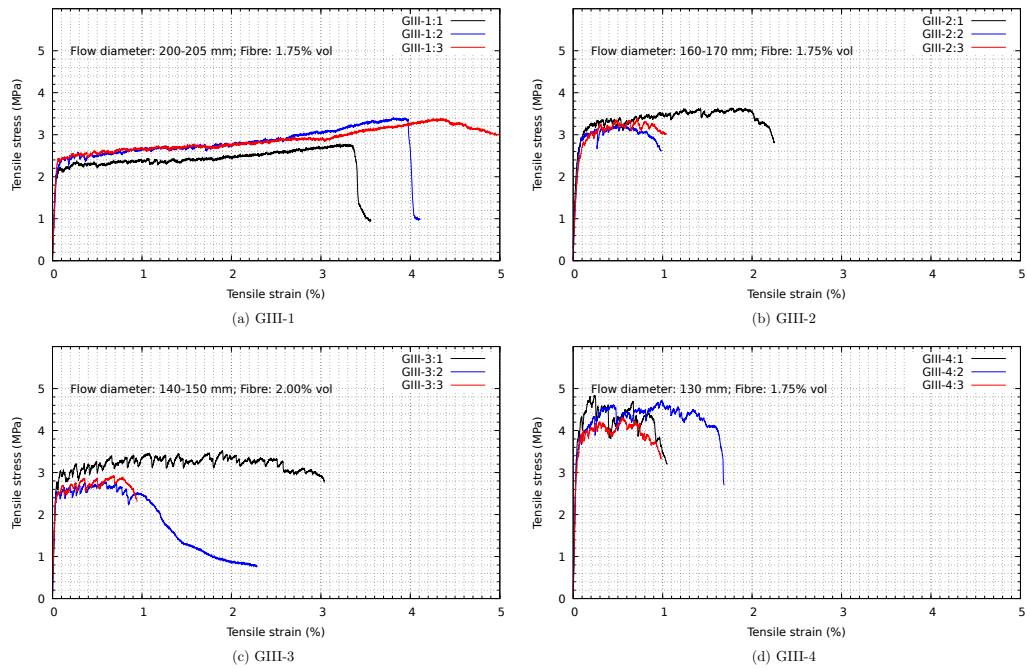


Figure 7: Results of tensile test of the lightweight composite of GIII.

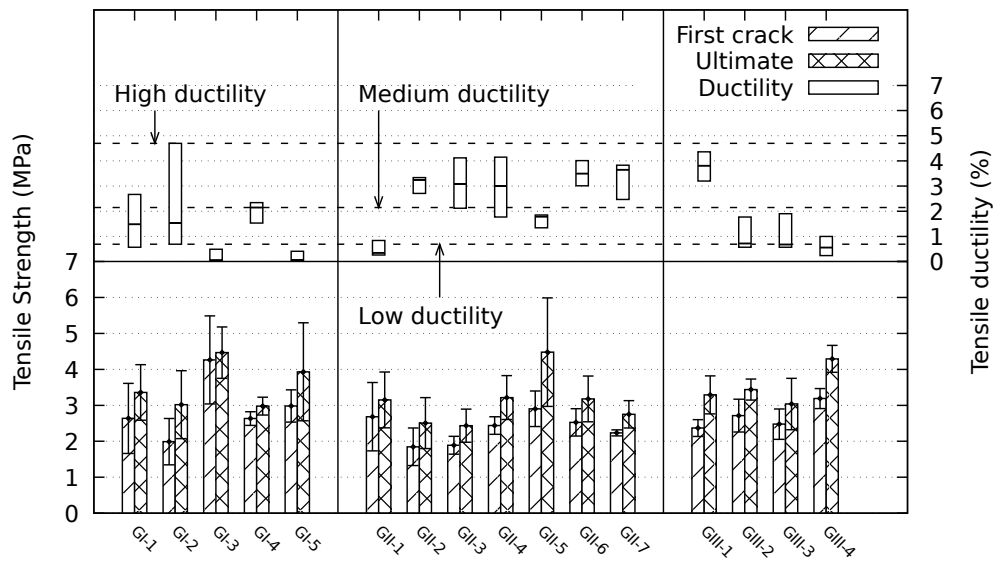


Figure 8: First crack strength, ultimate tensile strength and tensile ductility.

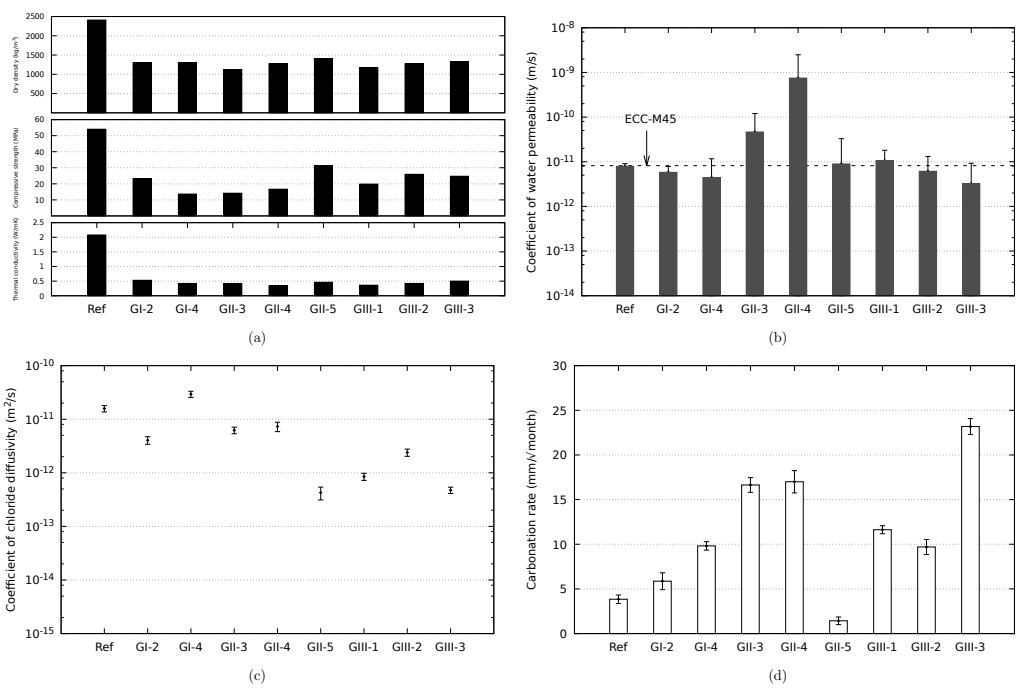


Figure 9: Results of (a) compression test, (b) thermal conductivity, (c) water permeability, (d) chloride diffusivity and (e) carbonation rate.

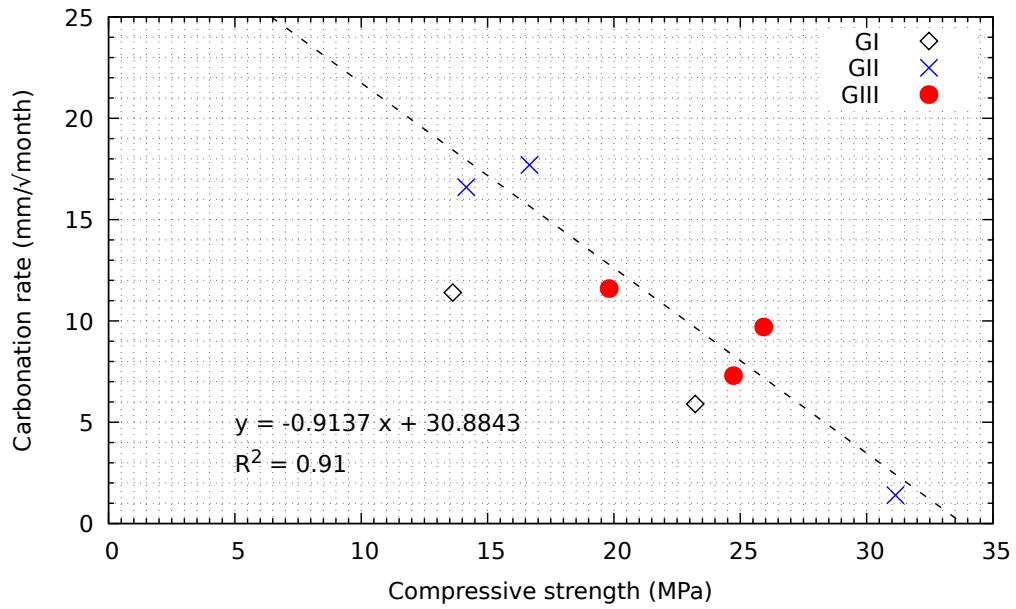


Figure 10: Relationship between carbonation rate and compressive strength of LSHCC. (The best-fitted line is only from the data of GII and GIII only.)

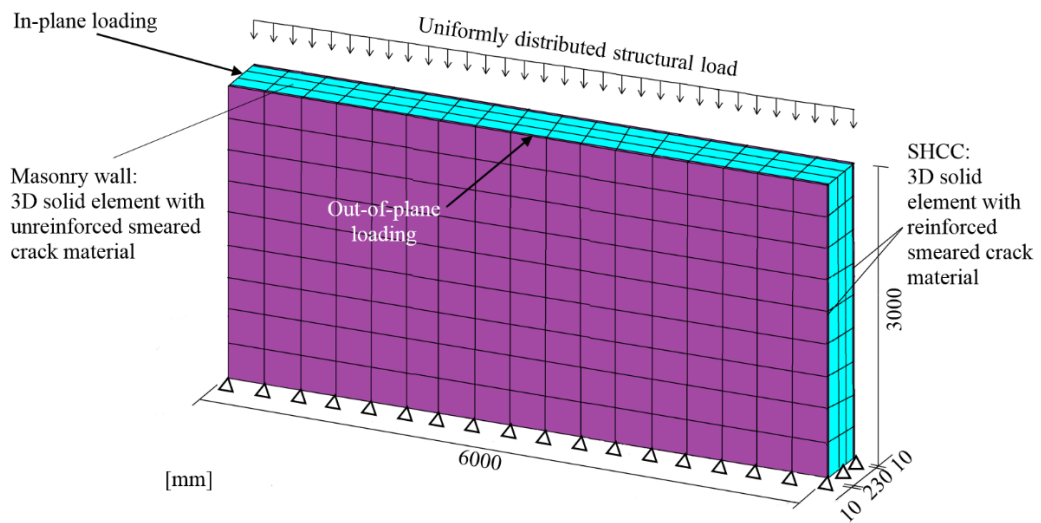
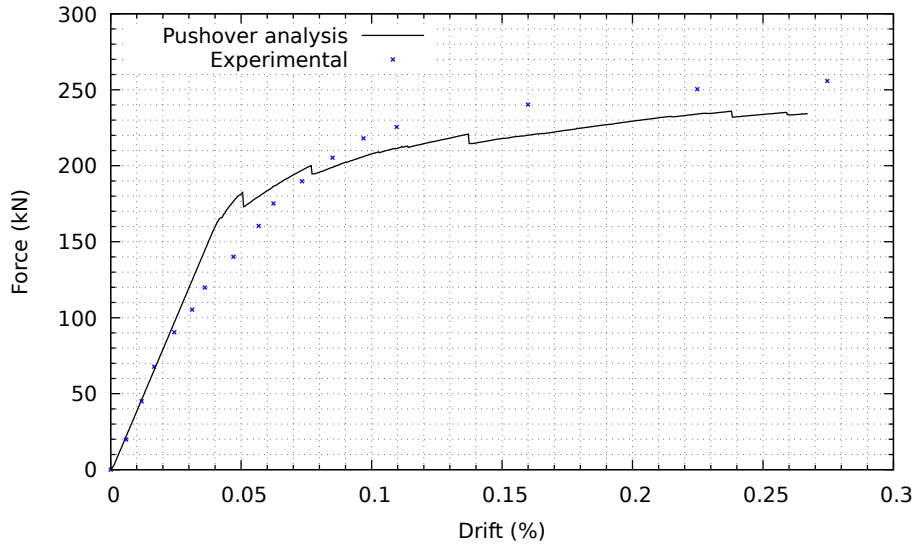
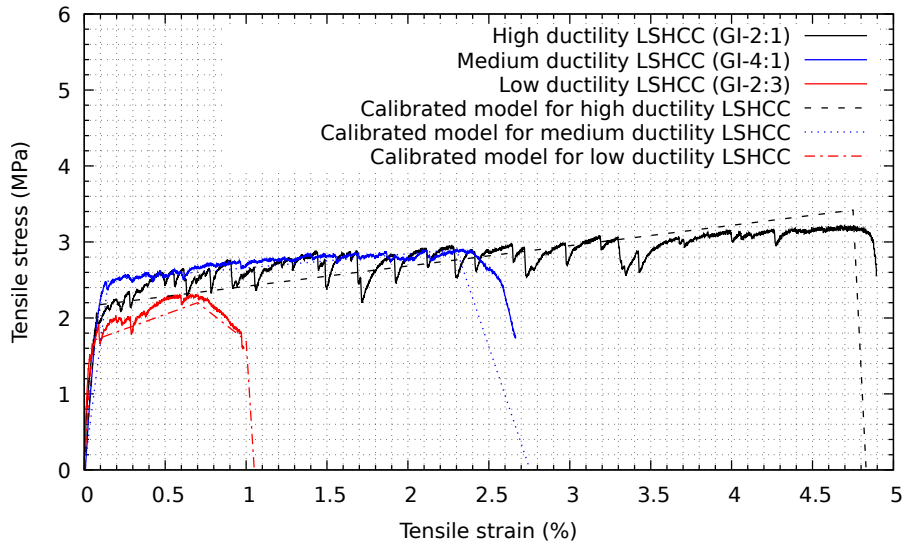


Figure 11: Three dimensional model of the pushover analysis.

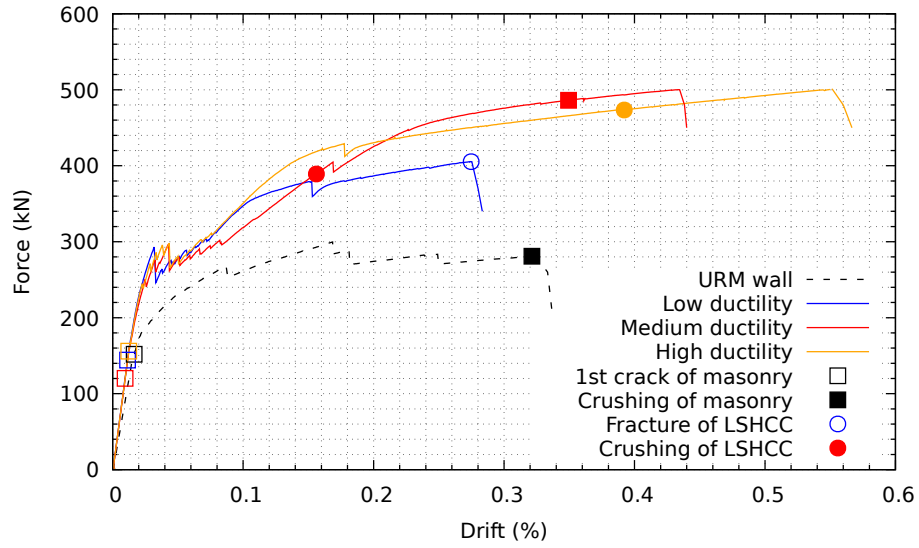


(a)

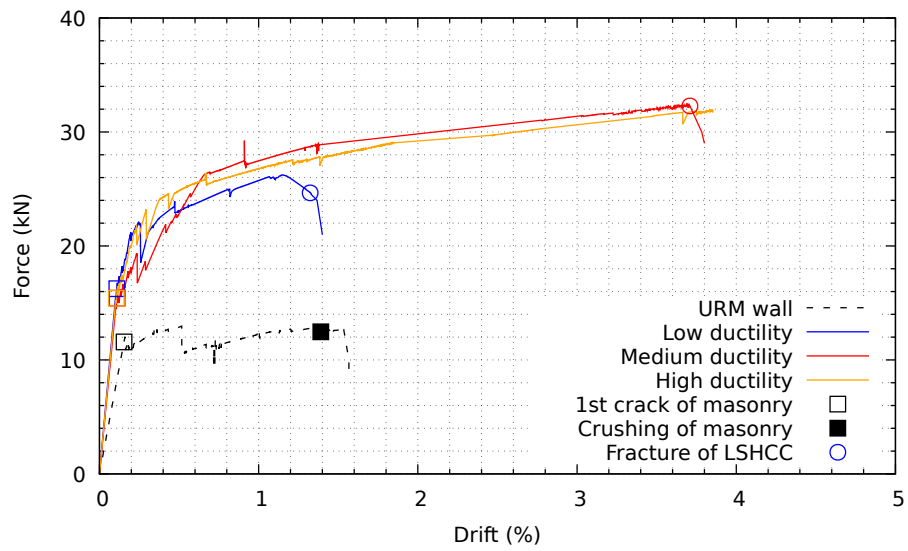


(b)

Figure 12: Model validation: (a) Masonry wall; (b) LSHCC..



(a)



(b)

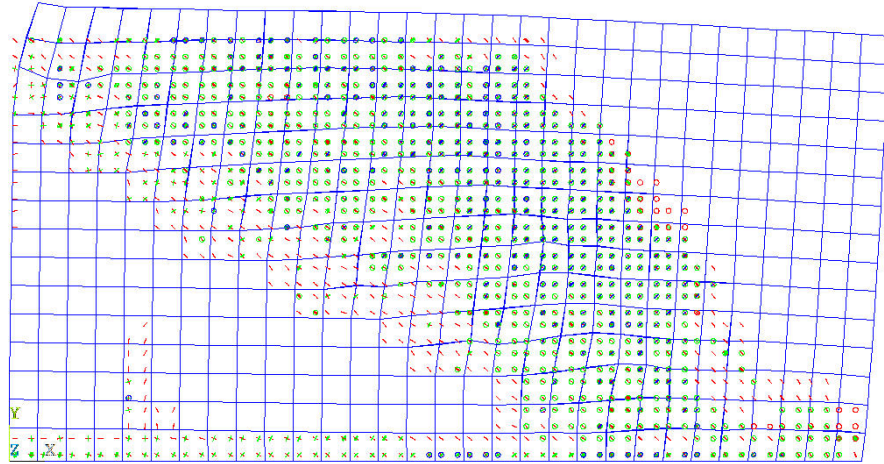
Figure 13: Base shear versus drift at the top of the wall: (a) In-plane; (b) Out-of-plane.

CRACKS AND CRUSHING

STEP=708

SUB =100

TIME=708



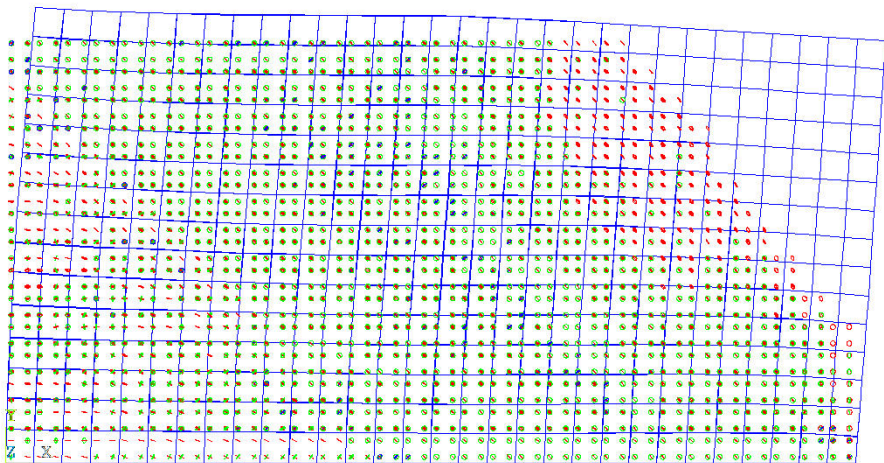
(a)

CRACKS AND CRUSHING

STEP=769

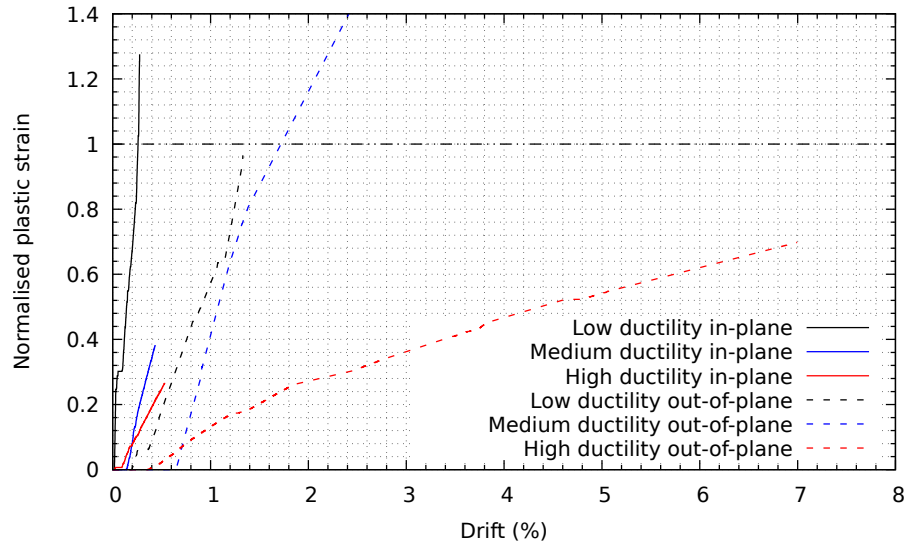
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TIME=769

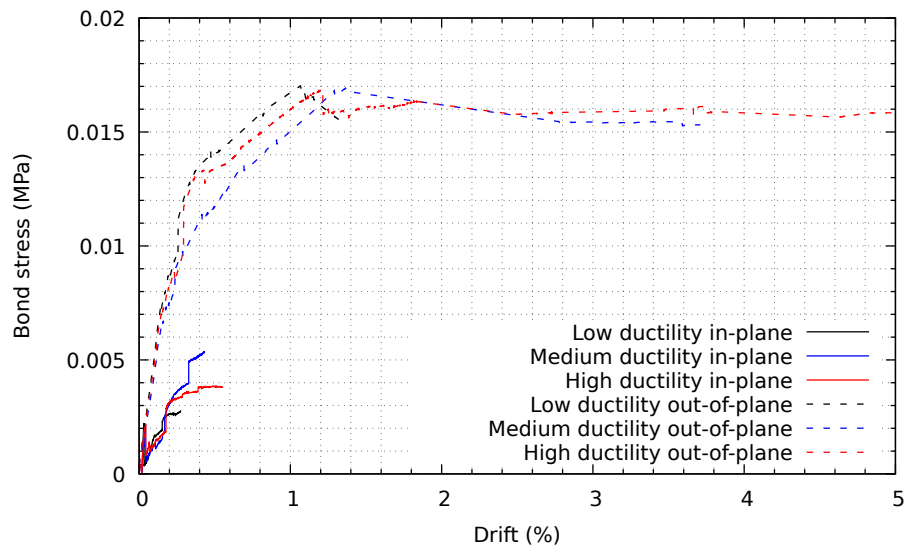


(b)

Figure 14: Crack patterns in the masonry wall element at 0.3% drift: (a) wall without LSHCC; (b) wall with medium ductility LSHCC



(a)



(b)

Figure 15: LSHCC response: (a) normalized plastic strain; (b) bond stress in the interface.

Table 1: Summary of raw materials, density, thermal conductivity of lightweight strain hardening composite.

	3M-S38	3M-S60	EG	MS
Matrix	OPC	OPC	FAG	FAG
LWA-matrix wt%	20	50	16	10
Density (kg/m ³)	1,450	930	1,754	1,586
Thermal conductivity (W/m·K)	N/A	N/A	~0.9	~1.1
Ultimate tensile strength (MPa)	4.31	2.85	3.8	3.4
Tensile ductility (%)	4.24	3.70	3.7	3.5

EG: expanded recycled glass

MS: microscopic hollow ceramic spheres

LWG: lightweight aggregates

FAG: fly ash based geopolymer

Table 2: XRF results of the raw materials of the cementitious matrix (in weight %).

	OPC	FA	GGBS
SiO ₂	19.4	52.0	32.2
CaO	67.0	4.7	46.5
Al ₂ O ₃	3.4	30.7	12.3
Fe ₂ O ₃	3.5	5.9	1.0
SO ₄	5.1	1.5	3.1
MgO	1.0	1.6	4.1
TiO ₂	0.2	2.3	0.6
MnO	0.2	0.1	0.2
K ₂ O	0.2	1.2	–

Table 3: Properties of PVA fibre.

Diameter	Length	Elastic modulus	Elongation	Nominal strength	Density
(μm)	(mm)	(GPa)	(%)	(GPa)	(kg/m^3)
39	12	41	6	1.6	1,300

Table 4: Properties of lightweight aggregates.

	unit	S15	S38*	S60*	EG [#]	MS [#]
Typical true specific gravity		0.15	0.38	0.6	1.4	0.85
Thermal conductivity	(W/m·K)	0.055	0.127	0.200	N/A	0.1
Particle size range	(μm)	25-90	15-75	15-55	40-125	38-125
Median particle size	(μm)	55	40	30	N/A	N/A
Isostatic crush strength	(MPa)	2.1	27.6	68.9	N/A	45

* is the glass micro-hollow bubble used in [32]

[#] is the expanded recycled glass and microscopic hollow ceramic spheres used in [43]

Table 5: Mix proportion of mortar test.

Mix	OPC	Water	Sand	S15	HPMC (%)	SP (%)	Dry mix (min)	Dry mix (speed)	Wet mix (min)	Wet mix (speed)	Water content (kg/m ³)	Estimated density (kg/m ³)	Measured density (kg/m ³)
A1	1	0.46	1	0.10	0.150	1.20	7	1	18	1	261	1,396	1,858
A2	1	0.46	1	0.10	0.150	1.20	7	1	8	2	261	1,396	1,924
A3	1	0.46	1	0.10	0.120	1.20	7	1	8	2	261	1,396	1,899
A4	1	0.46	1	0.10	0.120	1.00	7	1	18	1	261	1,396	1,858
A5	1	0.46	1	0.10	0.060	1.00	7	1	18	1	261	1,396	1,888
B1	1	0.50	1	0.10	0.225	1.70	7	1	14	1	278	1,388	1,604
B2	1	0.50	1	0.10	0.150	1.00	7	1	14	1	278	1,388	1,731
C1	1	0.55	1	0.10	0.188	1.00	7	1	10	1	297	1,377	1,487
C2	1	0.55	1	0.10	0.150	0.70	7	1	11	1	297	1,377	1,552
D1	1	0.60	1	0.10	0.188	0.63	7	1	9	1	316	1,368	1,417
D2	1	0.60	1	0.10	0.150	0.40	7	1	9	1	316	1,368	1,473
D3	1	0.60	1	0.10	0.113	0.30	7	1	9	1	316	1,368	1,523
D4	1	0.60	1	0.10	0.075	0.30	7	1	9	1	316	1,368	1,611
D5	1	0.61	1	0.10	0.150	0.20	7	1	10	1	316	1,366	1,728
D6	1	0.60	1	0.10	0.075	0.20	7	1	9	1	316	1,368	1,688
D7	1	0.60	1	0.10	0.000	0.20	7	1	8	1	316	1,368	1,623
D8	1	0.60	1	0.10	0.038	0.17	7	1	15	1	316	1,368	1,801
D9	1	0.60	1	0.10	0.000	0.15	7	1	13	1	316	1,368	1,747
D10	1	0.60	1	0.10	0.075	0.10	7	1	15	1	316	1,368	1,895
D11	1	0.60	1	0.10	0.000	0.10	7	1	18	1	316	1,368	1,861
E1	1	0.61	1	0.10	0.150	0.20	7	1	10	1	319	1,366	1,728
F1	1	0.71	1	0.10	0.150	0.15	7	1	10	1	353	1,349	1,584

Table 6: Mix proportion of direct tensile test.

Mix	Mix	OPC	FA	GGBS	Water	Sand	S15	HPMC	SP	Fibre	Water content (kg/m ³)	Estimated density (kg/m ³)	Measured density (kg/m ³)	Error %	Dry density (kg/m ³)	Flowability (mm)
								(%)	(%)	(% vol)						
48	GI-1	1	-	-	0.35	-	0.057	0.11	1.00	2.00	331	1,255	1,252	-0.2	1,245	160-170
54	GI-2*	1	-	-	0.4	0.165	0.060	0.15	1.00	2.00	327	1,378	1,309	-5.0	1,300	140-150
35	GI-3	1	-	-	0.375	0.165	0.065	0.12	1.20	1.75	304	1,345	1,420	5.6	1,350	120-130
51	GI-4*	1	-	-	0.47	0.500	0.080	0.11	1.50	2.00	300	1,360	1,338	-1.6	1,300	136-145
23	GI-5	1	-	-	0.35	0.165	0.065	0.15	1.50	1.75	290	1,364	1,415	3.7	1,392	120-130
5	GII-1	0.5	0.5	-	0.325	-	0.050	0.094	0.45	1.75	306	1,331	1,388	4.3	1,309	160
30	GII-2	0.2	0.8	-	0.375	0.165	0.050	0.11	0.70	1.75	309	1,350	1,297	-3.9	1,133	190-200
31	GII-3*	0.2	0.8	-	0.375	0.165	0.050	0.11	0.40	1.75	310	1,346	1,280	-4.9	1,119	190-200
17	GII-4*	0.2	0.8	-	0.35	0.165	0.050	0.11	0.70	1.75	295	1,356	1,379	1.7	1,277	180-185
11	GII-5*	0.67	0.33	-	0.35	0.165	0.060	0.11	0.80	1.75	291	1,350	1,433	6.1	1,401	140
26	GII-6	0.2	0.8	-	0.325	0.165	0.050	0.11	1.25	1.75	278	1,369	1,373	0.3	1,241	180-190
32	GII-7	0.2	0.8	-	0.325	0.165	0.050	0.11	1.00	1.50	280	1,366	1,354	-0.9	1,233	180-190
41	GIII-1*	0.2	0.6	0.2	0.375	0.165	0.052	0.11	0.40	1.75	311	1,354	1,266	-6.5	1,169	200-205
39	GIII-2*	0.2	0.2	0.6	0.375	0.165	0.060	0.11	0.70	1.75	307	1,347	1,356	0.7	1,274	160-170
52	GIII-3*	0.5	0.25	0.25	0.375	0.165	0.055	0.12	0.76	2.00	315	1,381	1,341	-2.9	1,328	140-150
42	GIII-4	0.5	-	0.5	0.375	0.165	0.052	0.11	0.80	1.75	311	1,355	1,457	7.5	1,356	130

*: transport properties were measured.

Table 7: Material properties of masonry wall model.

Elastic modulus (MPa)	Poissons ratio	Uniaxial crushing stress (MPa)	Uniaxial cracking stress (MPa)
2460	0.18	7.61	0.28